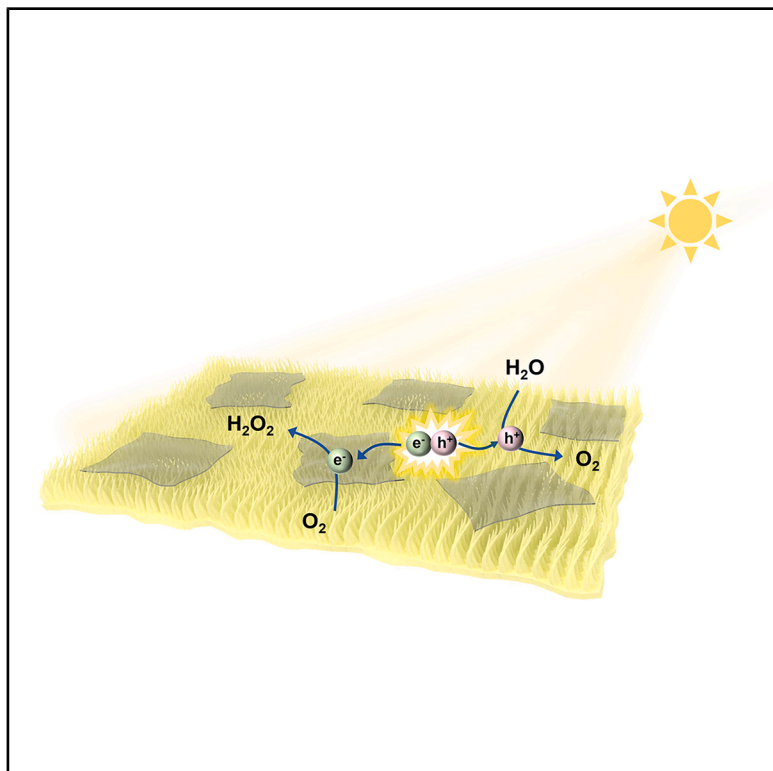


Self-floating COF-based Janus film: A robust triphase interface boosting charge separation for efficient H_2O_2 photosynthesis

Graphical abstract



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In brief

A self-floating COF-[NH₂-GO] Janus film was engineered via a one-step interfacial polymerization strategy for efficient H_2O_2 photosynthesis, in which the introduced graphene promoted the separation and transfer of photogenerated carriers. Meanwhile, the self-floating film photocatalysts enabled the stabilization of gas-liquid-solid triphase interface, significantly enhancing the O_2 mass transfer.

Highlights

- This COF-based Janus film boosts the photogenerated charge separation
- Self-floating film stabilizes the triphase interface for efficient mass transfer
- Photocatalytic film-device shows huge potential in practical applications
- This synthetic approach can be employed in various COFs and nanomaterials



Improvement

Enhanced performance with innovative design or material control

Li et al., 2026, Matter 9, 102679
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<https://doi.org/10.1016/j.matt.2026.102679>

Article

Self-floating COF-based Janus film: A robust triphase interface boosting charge separation for efficient H₂O₂ photosynthesis

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<https://doi.org/10.1016/j.matt.2026.102679>

PROGRESS AND POTENTIAL Artificial photosynthesis is a promising and sustainable approach for H₂O₂ production. As an emerging photocatalyst material, self-floating film, acting as a robust gas-liquid-solid triphase interface, can address the dilemma of sluggish O₂ mass transport in water for a triphase system. Nevertheless, the precise construction of heterostructures in films is still a huge challenge, causing the shortage of effective regulation over photogenerated charge kinetics. Herein, we present a self-floating COF-[NH₂-GO] film, synergistically optimizing charge kinetics and mass transport in triphase artificial photosynthesis. Hence, the photocatalytic production of H₂O₂ was greatly improved. Furthermore, a film-based device for *E. coli* sterilization was constructed, demonstrating its potential in practical applications. Importantly, this synthetic approach can be extended to other COFs, C₃N₄, and CNT, providing a universal blueprint for designing floating film photocatalyst for different demands.

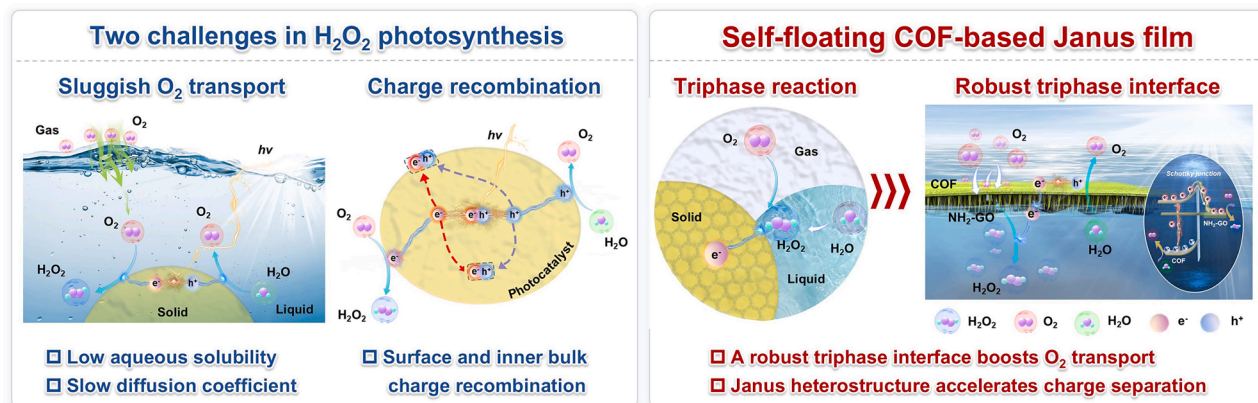
SUMMARY

H₂O₂ photosynthesis, as a gas-liquid-solid reaction, faces two giant challenges, i.e., charge recombination and sluggish oxygen transport. For improving efficiency, it is a feasible strategy to construct a robust triphase interface with optimized charge kinetics. Herein, we constructed a self-floating covalent-organic framework (COF)-based film with Janus heterostructure, i.e., COF-[NH₂-GO]. In overall H₂O₂ photosynthesis, COF-[NH₂-GO] exhibited the highest yield among COF film and powder, signifying improved separation and transfer of photogenerated carriers. The photocatalytic triphase system achieved 5,306.6 μmol·m⁻²·h⁻¹ H₂O₂ yield, higher than that of diphasic counterpart, ascribed to the promoted O₂ delivery. It was demonstrated that excellent photocatalytic performance was attributed to the improved charge kinetics and mass transfer. A prototypical film-based device was designed for photosynthesis and collection of H₂O₂, which exhibited bactericidal activity against *Escherichia coli*. More importantly, this strategy could be extended to other COFs and functional nanomaterials. This work provides new insights for challenging triphase photocatalysis.

INTRODUCTION

Hydrogen peroxide (H₂O₂), serving as a green oxidant and sustainable energy carrier, plays an indispensable role in chemical synthesis, wastewater treatment, medical disinfection, and other

fields.^{1–4} However, conventional industrial production methods (e.g., the anthraquinone process and thermal catalytic synthesis from H₂/O₂) suffer from intrinsic drawbacks, e.g., high energy consumption, complex processes, and safety risks.^{5–8} In contrast, artificial photosynthesis utilizes solar energy to directly



Scheme 1. Design principles and challenges for triphase photocatalytic H₂O₂ production

produce H₂O₂ from water and oxygen, and hence is regarded as an ideal and green approach owing to its eco-friendly advantage.^{9–13} Photocatalytic synthesis of H₂O₂ primarily proceeds through two fundamental pathways—the oxygen reduction reaction (ORR) and the water oxidation reaction (WOR).^{14–16} Due to the harsh conditions required for WOR, the two-electron ORR is currently one of central pathways for H₂O₂ photosynthesis.^{17–20} Essentially, ORR process is a triphase reaction, which involves gaseous O₂, liquid water, and solid photocatalysts. However, most of conventional photocatalytic systems are based on a liquid-solid diphasic system, where the photocatalytic activity is significantly limited by the low aqueous solubility (8 mg·L⁻¹ at 25°C) and a slow diffusion coefficient (2.10 × 10⁻⁹ m²·s⁻¹) of O₂ (scheme 1).²¹ In contrast, the O₂ diffusion rate in air of the gas-liquid-solid triphase system is approximately 10,000 times higher than that of a diphasic system, signifying more ample supply of O₂ for H₂O₂ production.²²

The superior O₂ mass transfer capability in triphase systems highlights the importance of gas-liquid-solid interface properties in photocatalytic H₂O₂ production. Various strategies have been developed to facilitate O₂ access to the catalyst surface (Figure S1). As one of the most common approaches, physical stirring was driven by external force to promote the contact between O₂ and catalyst, which requires continuous and extensive energy consumption.²³ In addition, light scattering in the suspension influences the light harvesting of photocatalysts, hindering the photocatalytic performance.²⁴ To circumvent these problems, some researchers have developed support photocatalysts on hydrophobic substrates to construct a triphase system. The stable gas-liquid-solid interface significantly accelerates gas diffusion and thereby boosts photocatalytic activity.^{25,26} However, these supported floating photocatalysts present several challenges, including catalyst detachment and blockage of active sites, which severely threaten both photocatalytic activity and stability.²⁷ Therefore, in order to address the aforementioned challenges, it is significant and urgent to develop self-floating film photocatalysts without additional substrates.

In recent decades, covalent organic frameworks (COFs), featuring high specific surface areas, regulable energy levels, tailorable frameworks, excellent stability, and pronounced light

responsiveness, have emerged as promising photocatalysts.^{28–36} Compared with powder samples, COF films possess some unique advantages such as low light scattering, excellent scalability, easy recyclability, facile device integration, and sufficient contact with reactants, making them a highly sought-after platform for artificial photosynthesis.^{37–39} Various strategies have been extensively investigated to construct COF films, e.g., interfacial polymerization,^{40,41} vapor-assisted conversion,⁴² and *in situ* solvothermal^{43,44} approaches. Among these, the liquid-liquid interfacial polymerization method has attracted significant attention due to its simple process and ability to produce high-quality films under mild conditions. The stable oil-water interface enables precise control over monomer diffusion and reaction kinetics, which is favorable for synthesizing continuous, uniform, and free-standing COF films.⁴⁵ Unfortunately, the photocatalytic performance of pristine COF films is limited by the rapid recombination of photogenerated electron-hole pairs and lack of active sites^{46,47} (scheme 1). Notably, as for powder photocatalysts, COF-based composite materials have been developed to address these dilemmas by constructing various heterojunctions, such as semiconductor-semiconductor^{48–50} and semiconductor-metal composites.⁵¹ Therefore, engineering heterostructures in COF films is regarded as a feasible and promising approach for enhancing the overall photocatalytic activity. However, the self-floating COF film with programmable heterojunctions is still rarely reported.

Herein, we present a one-step interfacial polymerization approach to construct a self-floating Janus film photocatalyst COF-[NH₂-GO] (scheme 1). Under pure water and open air without any sacrificial agents, H₂O₂ yield of COF-[NH₂-GO] film reached 922.10 μmol·g⁻¹·h⁻¹, which was around 1.44-fold higher than that of COF film (638.70 μmol·g⁻¹·h⁻¹) and 4.54-fold higher than that of COF powder (203.30 μmol·g⁻¹·h⁻¹). Based on spectral characterizations, it was demonstrated that the introduced graphene effectively facilitated the separation of photogenerated carriers, thereby enhancing the photocatalytic performance. Furthermore, the areal activity of COF-[NH₂-GO] film in the triphase system reached 3145.80 μmol·m⁻²·h⁻¹, representing a 2.36-fold enhancement compared with the diphasic system. Numerical simulations confirmed that a relatively high

O₂ concentration was always maintained at the triphase reaction interface, which was conducive to the generation of H₂O₂. Furthermore, a COF-[NH₂-GO] film-based device was constructed for H₂O₂ photosynthesis and collection, enabling effective bactericidal action against *Escherichia coli* (*E. coli*). The methodology has been successfully extended to diverse COFs and functional nanomaterials, all exhibiting substantially enhanced photocatalytic H₂O₂ production, confirming its broad applicability. To our knowledge, our work is the first example of a self-floating COF-based Janus film capable of achieving triphase photocatalytic H₂O₂ production, which not only advances the development of COF-based composite films but also offers a new design paradigm for photocatalytic H₂O₂ generation.

RESULTS AND DISCUSSION

Material synthesis and characterization

Owing to the extended π -conjugated framework, triazine-based COF materials exhibit remarkable visible light-harvesting capabilities.⁵² Besides, imine-linked COFs could be synthesized at room temperature.^{53,54} Therefore, we employed 2,4,6-tris(4-aminophenyl)-1,3,5-triazine (TTA) and 2,4,6-tris(4-formylphenyl)-1,3,5-triazine (TTB) as building blocks for the *in situ* construction of COF films at the liquid-liquid interface. TTB was dissolved in dichloromethane (DCM) as the oil phase at bottom. A small amount of water was then introduced on the oil surface as a buffer layer, into which an acetic acid aqueous solution containing TTA was carefully added. The as-prepared diphasic system was kept undisturbed at room temperature for 7 days, after which a yellow film formed at the liquid-liquid interface. After reaction, the upper aqueous layer was removed, and the synthesized COF film was collected and washed. As shown in Figure S2, the optical photograph showed that the COF film suspended on a metal wire loop was uniform and intact, signifying its self-supporting property. Scanning electron microscopy (SEM) images showed that the top surface (water phase) of the COF film was interweaved networks with elongated fibrous structures (Figure S3A). In contrast, the bottom surface of the film (oil phase) displayed a uniformly packed granular structure without obvious gaps (Figure S3B). The water contact angles of top and bottom surfaces of the COF film were 21.5° and 58.8°, respectively (Figure S4). The difference in surface wettability between the two sides of the COF film signified an asymmetric structure. Cross-sectional SEM image indicated a uniform thickness of approximately 8 μ m (Figure S3C). High-resolution transmission electron microscopy (HRTEM) images revealed clear lattice fringes of the COF film with a spacing of approximately 2.0 nm, corresponding to the (100) crystal facet, which indicated good crystallinity (Figure S5). To further determine the crystalline structure, X-ray diffraction (XRD) was conducted on the COF film. The characterized peaks at 4.0°, 7.1°, 8.2°, and 10.8° corresponded to the (100), (110), (200), and (120) crystal faces, respectively, which were consistent with those of the simulated one (Figure S6). For comparison, the corresponding COF powder was synthesized via a solvothermal method. XRD and Fourier transform infrared (FTIR) spectroscopy confirmed the successful preparation of COF powder (Figures S6 and S7). SEM image of the powder sample showed a loose morphology,

while HRTEM observations clearly indicated lattice fringes (Figures S8 and S9).

Given that the charge separation efficiency plays an important role in photocatalytic performance, graphene with superior electrical conductivity was selected as the functional unit to be integrated into COF films. In order to optimize the interaction between COF and graphene, NH₂ groups were grafted on the surface of graphene oxide to prepare NH₂-functionalized graphene oxide (NH₂-GO), via a covalent bonding strategy between anchoring (3-aminopropyl) triethoxysilane (APTS) and graphene oxide (GO) (Figure S10). Compared with GO, FTIR spectra of NH₂-GO (Figure S11) showed new peaks at 2951, 1634, and 1125 cm⁻¹, assigned to the stretching vibrations of C-H, N-H, and Si-O-C bonds, respectively, which confirmed the successful incorporation of APTS onto GO. Transmission electron microscope (TEM) image showed the maintained nanosheet morphology of NH₂-GO (Figure S12A). Furthermore, energy dispersive X-ray spectroscopy (EDS) elemental mapping of NH₂-GO showed a uniform distribution of C, N, O, and Si (Figures S12B–S12F). These results confirmed that the surface of GO was successfully decorated by amino functional groups.

The as-prepared NH₂-GO was employed in the liquid-liquid interfacial approach to directly construct COF-[NH₂-GO] composite film. This was achieved by simply replacing the buffer water layer with an aqueous dispersion of NH₂-GO, while keeping all other steps unchanged (Figure 1A). The as-prepared COF-[NH₂-GO] film exhibited excellent and stable self-floating ability, which was crucial for stabilizing the gas-liquid-solid triphase interface during photocatalysis. As shown in Figures 1B and S13, the film could stably float on water. Under static conditions, this composite film could float on water surface for over one month without sinking. As shown in Figure S14 and Videos S1 and S2, this self-floating COF-[NH₂-GO] film could remain stable on the water surface, even under vigorous stirring (400 rpm), which was closely correlated with the film's excellent mechanical properties and hydrophobicity. As shown in Figure 1C, the continuous and intact film could be transferred onto a metal wire loop, demonstrating its superior self-supporting property. Moreover, the stable liquid-liquid interface guaranteed the large-scale synthesis of COF and COF-[NH₂-GO] films with a lateral size of 5 cm (Figure S15). The surface morphology and microstructure of the COF-[NH₂-GO] film was further investigated by SEM. The SEM images showed that on the top surface, two-dimensional graphene sheets were uniformly distributed and intertwined with the fibrous network of COF, forming a tightly integrated structure (Figures 1A and S16A). This morphological feature was further corroborated by the cross-sectional SEM image (Figure 1A), confirming the successful incorporation of NH₂-GO and its strong interfacial integration into the COF matrix. In contrast, the bottom surface of the COF-[NH₂-GO] film consisted of uniformly packed granular architecture similar to the COF film (Figure S16B). Notably, the COF-[NH₂-GO] film demonstrated enhanced hydrophilicity, with the decrease in top surface contact angle to 12° (Figure S17). Overall, it was demonstrated that the COF-[NH₂-GO] film possessed an asymmetric Janus-type structure.

To further investigate the film crystallinity, XRD and HRTEM were employed to characterize the COF-[NH₂-GO] Janus film. As

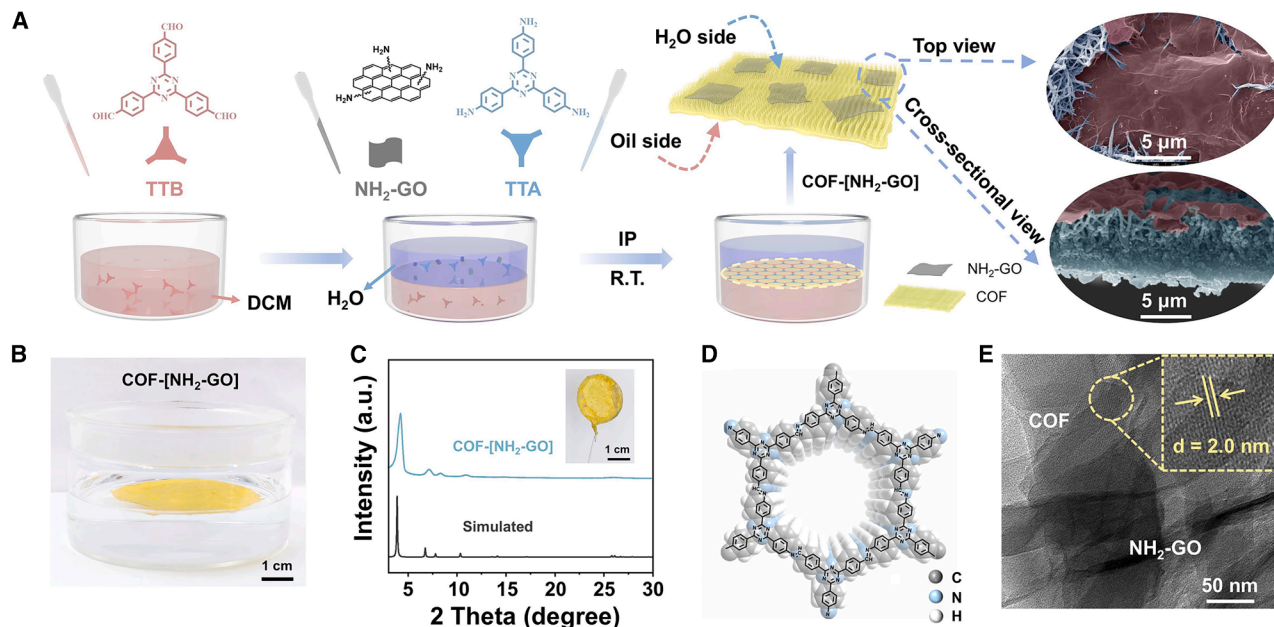


Figure 1. Synthesis and characterizations of the COF-[NH₂-GO] Janus film

(A) Synthesis process of the COF-[NH₂-GO] film, and SEM images of the top surface (in H₂O phase) and cross-section of the COF-[NH₂-GO] film (DCM: dichloromethane; GO: graphene oxide; TTB, 2,4,6-tris(4-formylphenyl)-1,3,5-triazine; and TTA, 2,4,6-tris(4-aminophenyl)-1,3,5-triazine).

(B) Optical photograph of as-prepared COF-[NH₂-GO] film floating on the water surface. Scale bar, 1 cm.

(C) XRD profiles of the COF-[NH₂-GO] film and the simulation result.

(D) The structure of COF based on TTB and TTA.

(E) HRTEM image of the COF-[NH₂-GO] film.

XRD, X-ray diffraction; HRTEM, high-resolution transmission electron microscopy.

shown in Figures 1C and 1D, the XRD pattern of COF-[NH₂-GO] composite film exhibited characteristic peaks corresponding to the COF component, confirming the crystalline structure of the COF within composite films. HRTEM image of the COF-[NH₂-GO] film not only revealed the integration of COF and NH₂-GO but also showed that the crystal fringe of COF was approximately 2.0 nm (Figure 1E). This spacing corresponded to the (100) crystal facet and was consistent with that of the COF film, further verifying the crystalline structure of COF in the composite films. In addition, EDS elemental mapping of COF-[NH₂-GO] film showed the uniform distribution of C, N, O, and Si (Figure S18), further revealing the successful integration COF and NH₂-GO in the composite films. FTIR spectra of the COF-[NH₂-GO] film exhibited a new characteristic peak of imine stretching vibration at 1,628 cm⁻¹, confirming the formation of imine bond within the composite material (Figure S19). The porosity of COF-[NH₂-GO] and COF films was evaluated by N₂ sorption isotherms measured at 77 K. As shown in Figure S20, the specific surface area of the COF film was 749.89 m²·g⁻¹, and after the incorporation of NH₂-GO, the specific surface area of COF-[NH₂-GO] film slightly decreased to 530.18 m²·g⁻¹. Thermogravimetric analysis (TG) results demonstrated that both COF and COF-[NH₂-GO] films possessed excellent thermal stability (Figure S21). The above-mentioned results all demonstrated the successful preparation of COF-[NH₂-GO] Janus film.

As a kind of semiconductor material, the band structure of COF should be determined. UV-Vis diffuse reflectance spectroscopy (UV-vis DRS) was employed to characterize the light capturing

ability of COF. As shown in Figure 2A, the COF film exhibited pronounced absorption in the visible region. The incorporation of NH₂-GO significantly enhanced the absorption intensity and induced a distinct red-shift in the absorption edge, indicating improved visible light utilization. The optical band gap (E_g) of COF was determined to be 2.45 eV by Tauc plot analysis (Figure S22A), while the Mott-Schottky measurement result (Figure S22B) showed that the conduction band (CB) position of COF was -0.76 V relative to normal hydrogen electrode (NHE). Afterward, the valence band (VB) position was calculated to be 1.69 V. This band structure of COF film was thermodynamically feasible for the photocatalytic reduction of O₂ to H₂O₂ and the oxidation of H₂O to O₂. Furthermore, ultraviolet photoelectron spectroscopy (UPS) measurements determined work function values of 5.35 eV and 5.47 eV for COF and NH₂-GO, respectively (Figure S23). The higher Fermi level of COF relative to NH₂-GO prompted electron transfer from COF to NH₂-GO upon contact until equilibrium was reached. Within this configuration, photoexcited electrons were preferentially injected into the conductive NH₂-GO domains, which served as electron collectors and facilitated the adsorption and activation of O₂ molecules. Thus, NH₂-GO could act as a charge mediator to promote charge separation and interfacial electron transfer from COF to NH₂-GO (Figure 2B).

Photocatalytic H₂O₂ performance

Given the light-capturing capability and band structures of COF and graphene, the COF and COF-[NH₂-GO] Janus films were

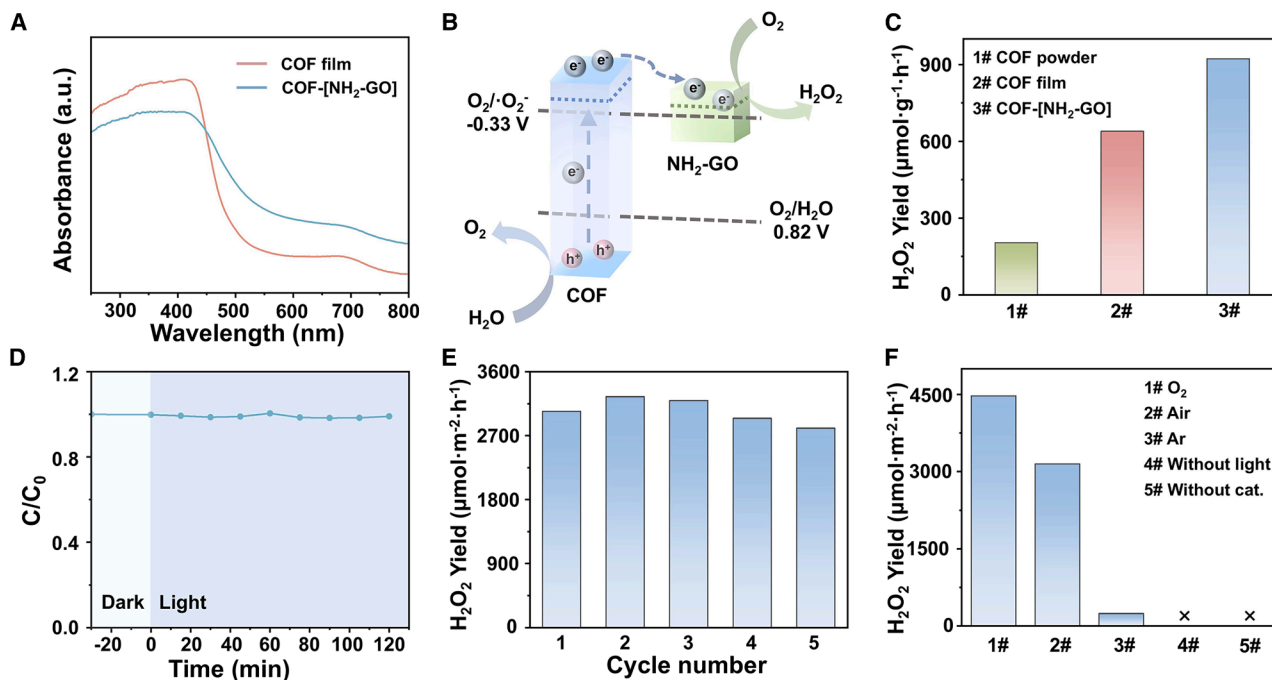


Figure 2. Photocatalytic experiments for powder and film samples

- (A) UV-vis spectra of the COF and COF-[NH₂-GO] films.
 (B) Band structure of the COF-[NH₂-GO] film.
 (C) Photocatalytic mass-activity of COF powder, COF film, and COF-[NH₂-GO] film for the photosynthesis of H₂O₂.
 (D) H₂O₂ (100 μM) degradation study for the COF-[NH₂-GO] film under dark and illumination in Ar atmosphere.
 (E) Cycling experiments of the COF-[NH₂-GO] film for the photosynthesis of H₂O₂.
 (F) Control experiments of the COF-[NH₂-GO] film for the photosynthesis of H₂O₂.

placed in an air-water system to evaluate their photocatalytic performance for H₂O₂ synthesis under visible light ($\lambda > 420$ nm, 100 mW cm⁻²) without any sacrificial agents. The concentration of the generated H₂O₂ was quantified using iodometry colorimetry (Figure S24). As shown in Figure 2C and Table S1, the photocatalytic H₂O₂ production rate over COF-[NH₂-GO] film reached 922.10 μmol·g⁻¹·h⁻¹, which was 1.44- and 4.54-fold higher than that of COF film and COF powder, respectively. This enhancement suggested that the incorporation of GO facilitated the separation and transfer of photogenerated carriers, thereby improving the photocatalytic activity for H₂O₂ synthesis. Notably, the areal activity of COF-[NH₂-GO] film attained 3145.80 μmol·m⁻²·h⁻¹. In comparison with previous COF-based photocatalysts, the as-synthesized COF-[NH₂-GO] film exhibited excellent photocatalytic activity (Table S2). Due to the poor stability of H₂O₂, the degradation in the presence of the COF-[NH₂-GO] and COF films was investigated under both dark and illuminated conditions. As shown in Figure 2D, the concentration of H₂O₂ remained nearly constant under both dark and illuminated conditions, indicating that neither the presence of the catalyst nor the illumination process exerted a significant effect on the decomposition of H₂O₂. Furthermore, the COF-[NH₂-GO] film demonstrated excellent stability in the photocatalytic production of H₂O₂. Even after five consecutive catalytic cycles, the yield of H₂O₂ showed no significant decline (Figure 2E; Table S3), which attested to the high photostability and reusability of the catalyst. After the photocatalytic

reaction, the COF-[NH₂-GO] film still floated on water surface and maintained whole integrity without cracking (Figure S25), certifying the stable gas-solid-liquid triphase interface. In addition, the used COF-[NH₂-GO] film was further confirmed by XRD, HRTEM, FTIR, and SEM images (Figures S26–S29), revealing that its crystalline phase, composition, and morphology could be maintained after the photocatalytic process. Furthermore, the photocatalytic performance of the COF-[NH₂-GO] film for H₂O₂ production was evaluated under natural sunlight. As shown in Figure S30 and Table S4, the H₂O₂ yield achieved approximately 2,500.00 μmol·m⁻²·h⁻¹ under solar illumination. Furthermore, the COF-[NH₂-GO] film showed excellent photocatalytic H₂O₂ yields in lake water (2931.1 μmol·m⁻²·h⁻¹) and seawater (2850.6 μmol·m⁻²·h⁻¹), as shown in Figure S31 and Table S5. These results further demonstrated the practical applicability of the COF-[NH₂-GO] film and underscored its potential for practical solar-driven synthesis of H₂O₂. The photocatalytic generation of H₂O₂ over the COF-[NH₂-GO] film was explored through a series of control experiments to investigate the reaction process of H₂O₂ production. As shown in Figure 2F and Table S6, H₂O₂ formation was not observed in the absence of either COF-[NH₂-GO] film or light irradiation, confirming that both the photocatalyst and light were essential for the reaction. H₂O₂ was hardly detectable under an argon atmosphere, while an oxygen atmosphere led to a higher yield compared to air, indicating that oxygen participated in H₂O₂ photosynthesis.

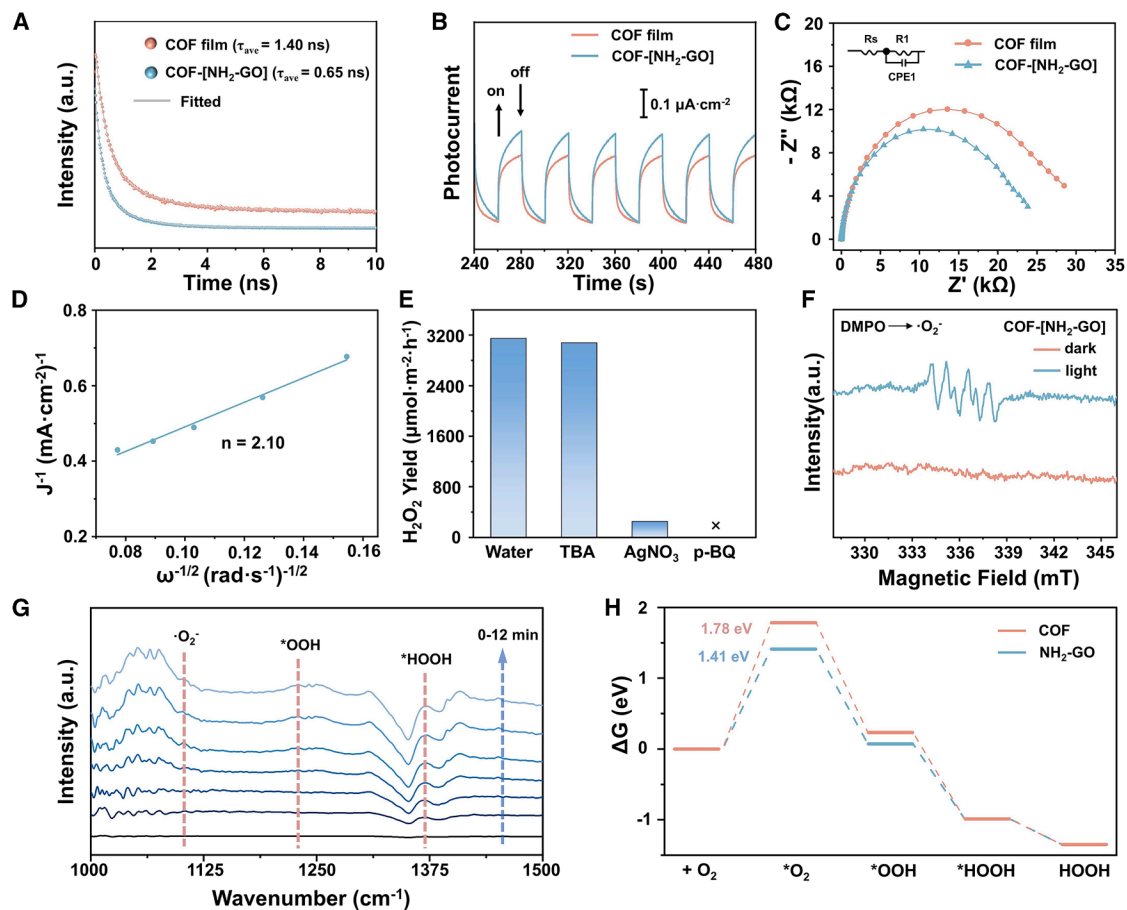


Figure 3. Mechanistic investigation for photocatalytic H₂O₂ production

(A–C) TRPL spectra (excitation wavelength: 375 nm), photocurrent measurements, and EIS Nyquist plots of the COF and COF-[NH₂-GO] films. (D) Koutecky-Levich plot of the COF-[NH₂-GO] film obtained via RDE measurement in phosphate buffer solution (pH = 7) with continuous O₂ purging. (E) Photocatalytic activity of COF-[NH₂-GO] film for synthesizing H₂O₂ in the presence of TBA (10 mM), AgNO₃ (10 mM), or p-BQ (10 mM). (F) EPR spectra of COF-[NH₂-GO] film in the presence of DMPO. (G) *In situ* FTIR spectra of COF-[NH₂-GO] film in H₂O₂ photosynthesis. (H) Free energy diagrams of photocatalytic O₂ reduction pathways on COF and NH₂-GO within COF-[NH₂-GO] film. TRPL, time-resolved photoluminescence; RDE, rotating disk electrode; EPR, electron paramagnetic resonance; and FTIR, Fourier transform infrared.

The exceptional photocatalytic performance of the COF-[NH₂-GO] film in H₂O₂ photosynthesis prompted a detailed investigation into the kinetics of photogenerated carriers. Photoluminescence (PL) spectroscopy showed that the fluorescence intensity of COF-[NH₂-GO] was lower than that of COF (Figure S32), indicating that the recombination of photogenerated electrons and holes was significantly suppressed. In addition, time-resolved photoluminescence (TRPL) spectroscopy revealed that the average exciton lifetime of the COF-[NH₂-GO] film was 0.65 ns, lower than that of the COF film (1.40 ns), suggesting that the introduced NH₂-GO significantly accelerated charge transfer (Figure 3A and Table S7). In the photocurrent measurement, the photocurrent density of the COF-[NH₂-GO] film was higher than that of the COF film (Figure 3B), indicating a faster charge transfer rate in the COF-[NH₂-GO] film. Moreover, electrochemical impedance spectroscopy (EIS) was employed to evaluate the charge kinetics. As

shown in Figure 3C, the semicircle radius in the Nyquist plot of the COF-[NH₂-GO] film was significantly smaller than that of the COF film, which indicated that the grafting of graphene onto COF indeed promoted charge transfer. In summary, all the above lines of evidence suggested that the introduction of graphene helped improve the separation of photogenerated electrons and holes, thereby achieving more efficient photocatalysis.

Photocatalytic mechanism investigation

To elucidate the mechanism of photocatalytic H₂O₂ production, linear sweep voltammetry (LSV) curves were measured using a rotating disk electrode (RDE), and the electron transfer number (*n*) involved in the ORR was calculated. Based on the LSV results (Figures 3D, S33, and S34), the Koutecky-Levich (K-L) plots yielded *n* values of 2.1 and 1.5 for the COF-[NH₂-GO] film and the COF film, respectively. It was indicated that the ORR process over these two films followed a two-electron reduction pathway.

Moreover, the n value of the COF-[NH₂-GO] film was closer to 2 compared with that of the COF film, suggesting improved selectivity toward the two-electron O₂ reduction after the introduction of GO units. In order to further clarify the reaction pathway and identify the active species involved in the 2e⁻ ORR process, quenching experiments were performed using silver nitrate (AgNO₃), tert-butano (TBA), and *p*-benzoquinone (*p*-BQ) as scavengers for photogenerated electrons (e⁻), hydroxyl radicals (·OH), and superoxide radicals (·O₂⁻), respectively (Figure 3E and Table S8). The introduction of AgNO₃ resulted in a significant suppression of the H₂O₂ production rate, confirming that photogenerated electrons were essential for reducing O₂ to H₂O₂. In contrast, the presence of TBA caused only a marginal variation in the H₂O₂ yield, ruling out the involvement of ·OH radicals. Most notably, the addition of *p*-BQ led to a drastic decline in H₂O₂ formation, highlighting the critical role of ·O₂⁻ as a key intermediate and corroborating the 2e⁻ two-step ORR mechanism. Furthermore, electron paramagnetic resonance (EPR) spectroscopy was employed with 5,5-dimethyl-1-pyrroline N-oxide (DMPO) as a spin-trapping agent to verify the presence of ·O₂⁻. Under light irradiation, the EPR spectra of the COF-[NH₂-GO] film exhibited distinct DMPO-·O₂⁻ signals (Figure 3F), further supporting that the system predominantly followed an indirect 2e⁻ two-step ORR pathway. It was worth noting that the COF-[NH₂-GO] film produced more ·O₂⁻ than the COF film (Figures 3F and S35), indicating more efficient charge separation and transfer within the composite film. Additionally, an isotopic labeling experiment was conducted using H₂¹⁸O ($m/z = 20$), instead of H₂O, and the products were analyzed by mass spectrometry. As shown in Figure S36, the peak observed at $m/z = 36$ could be attributed to ¹⁸O₂, confirming that the evolved O₂ originated from water oxidation. This result indicated that water was preferentially oxidized to oxygen via a four-electron water oxidation pathway (2H₂O → O₂ + 4H⁺ + 4e⁻) in the COF-[NH₂-GO] film.

In situ FTIR measurement was conducted to monitor the key intermediate species during the ORR process. As illustrated in Figure 3G, the characteristic peaks of ·O₂⁻ (1106 cm⁻¹), *OOH (1229 cm⁻¹), and *HOOH (1370 cm⁻¹) were observed, and their intensities increased gradually with prolonged irradiation time. The detection of ·O₂⁻ intermediate further confirmed that H₂O₂ production over the COF-[NH₂-GO] film occurred via the indirect 2e⁻ two-step ORR mechanism. In addition, density functional theory (DFT) calculations were performed to evaluate free energy changes of the photocatalytic 2e⁻ ORR on both COF and NH₂-GO, aiming to identify the preferential sites for O₂ reduction within the composite material (Figures S37 and S38; Table S16). The reaction pathways were assessed based on four essential steps toward H₂O₂ formation: initial oxygen adsorption onto the catalyst surface forming *O₂, the protonation of *O₂ to generate *OOH, further protonation of *OOH to form *HOOH, and *HOOH desorption from the catalyst to yield H₂O₂. As shown in Figure 3H, NH₂-GO exhibited a lower free energy change (ΔG) for the formation of *O₂ (1.41 eV) than COF (1.78 eV), demonstrating its superior capability in oxygen adsorption and activation. Furthermore, the ΔG value associated with the conversion of *O₂ to *OOH was also lower on NH₂-GO, which facilitated the progression of subsequent reaction steps. Therefore, the photocatalytic ORR for H₂O₂ generation that

occurs on NH₂-GO was thermodynamically more favorable. This result is consistent with those of previously published studies.^{55,56}

Numerical simulations

The photocatalytic production of H₂O₂ is a complicated process at the gas-liquid-solid triphase interface, wherein the oxygen delivery is an important limiting factor. The as-synthesized floating film could effectively stabilize the gas-liquid-solid triphase interface as the catalyzing platform for H₂O₂ photosynthesis. To demonstrate the superiority of the triphase system, the photocatalytic performances of the films were evaluated in diphase and triphase systems. As shown in Figure 4A, the COF-[NH₂-GO] Janus film submerged at the bottom of the reactor was recognized as a diphase system, whereas the COF-[NH₂-GO] film floating on water surface was defined as a triphase system. The photocatalytic reaction was carried out in pure water under visible light irradiation, using air as the oxygen source. As shown in Figure 4B and Table S9, the H₂O₂ production rate in the triphase system reached 3145.80 μmol·m⁻²·h⁻¹, approximately 2.36-fold higher than in the diphase system (1332.79 μmol·m⁻²·h⁻¹). This remarkable enhancement in performance could be attributed to the more efficient oxygen mass transport in the triphase system.

At this triphase interface, the local O₂ concentration around the photocatalyst played an important role in optimizing the photocatalytic performance. We conducted numerical simulations using the finite-element method (FEM) and physics-informed neural networks (PINNs) to model the mass transport and concentration distributions for both the diphase and triphase systems. The two approaches gave almost identical results (Tables S10 and S11). In the diphase system, the catalytic layer was situated between two aqueous phases (each 5-mm thick), whereas in the triphase system, it was placed between a gas phase (5 mm) and an aqueous phase (5 mm). Considering oxygen solubility under standard conditions (25°C, 101.3 kPa), the bulk O₂ concentrations for the gas and aqueous phases were set to 8.59 mM and 0.27 mM, respectively. The simulations compared the oxygen transport flux (J_{O_2}) at the interface and visualized the resulting steady-state spatial distributions of O₂ and H₂O₂ (Figures 4C–4F). The calculated oxygen fluxes provided direct evidence for the advantage in mass transport for the triphase system. As shown in Figure 4E, the total O₂ flux in the triphase system reached 4.500×10^{-7} mol m⁻² s⁻¹, significantly higher than that in the diphase system (1.848×10^{-7} mol m⁻² s⁻¹), with the O₂ flux coming predominantly from the upper surface (the gas phase). This stark contrast unequivocally demonstrated that the direct supply of gaseous O₂ was the dominant and highly efficient pathway for oxygen delivery to the catalytic interface. The different oxygen transport capabilities govern the interfacial concentration distributions (Figures 4C, 4D, and 4F). The simulation results revealed that for the diphase system, a significant depletion of O₂ could be noted near the interface, owing to the limited aqueous diffusion flux. In contrast, for the triphase system, the high flux from the gas phase enabled efficient and continuous replenishment of O₂, thereby maintaining a high concentration at the reaction interface, close to its saturation solubility in water. Furthermore,

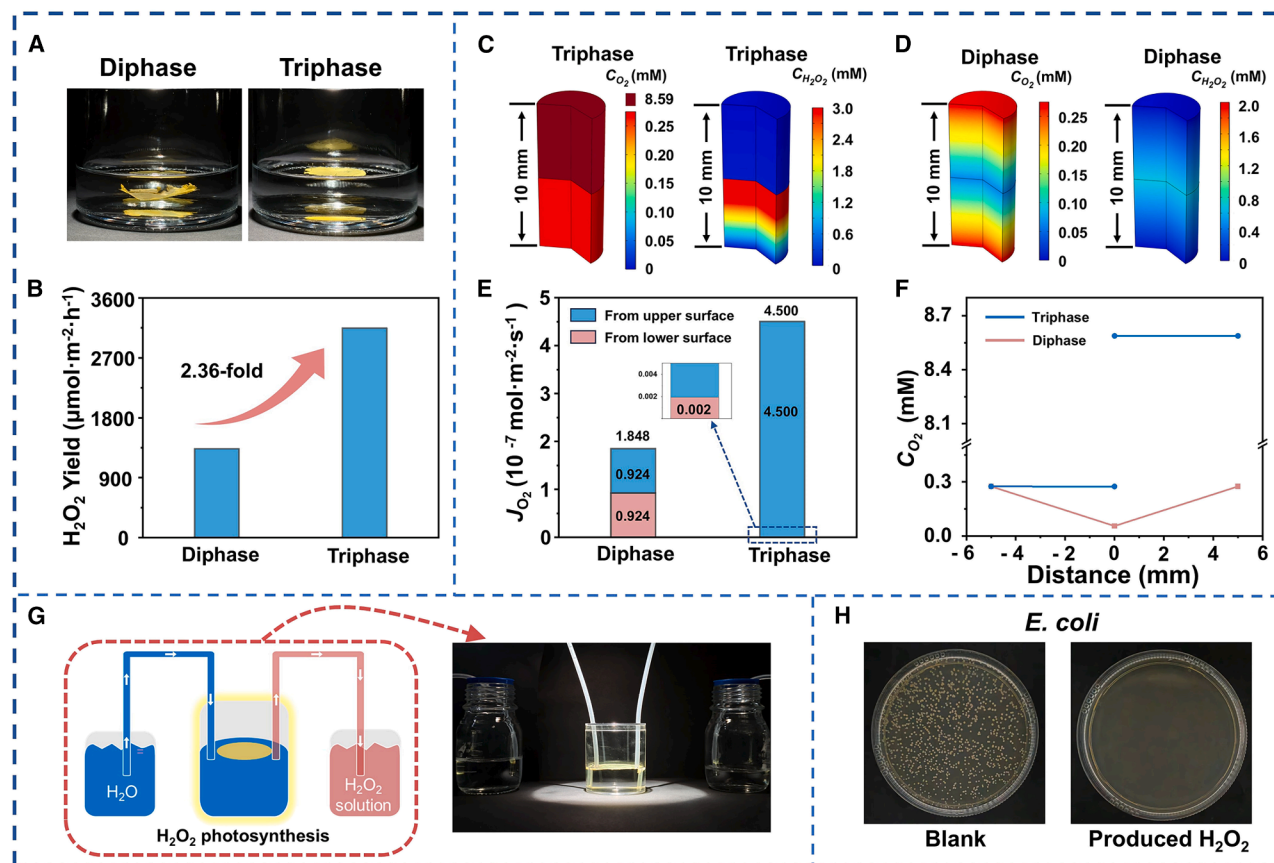


Figure 4. Numerical simulation of the triphase photocatalysis system and practical application

(A) Photographs of the liquid-solid diphasic system and the gas-liquid-solid triphasic system for the COF-[NH₂-GO] film.

(B) Comparison in terms of photocatalytic activity for the diphasic and triphasic systems for the COF-[NH₂-GO] film.

(C and D) Comparisons in terms of steady-state distribution of O₂ and H₂O₂ concentrations. (C) Comparison of O₂ and H₂O₂ concentrations for the triphasic system. (D) Comparison of O₂ and H₂O₂ concentrations for the diphasic system.

(E) Comparison in terms of oxygen transport flux of the triphasic and diphasic systems.

(F) Oxygen concentration in the triphasic and diphasic systems, calculated from the diffusion model.

(G) Schematic of the prototypical film-based device for photosynthesis and collection of H₂O₂.

(H) Sterilization of *E. coli* by photosynthesized H₂O₂ over COF-[NH₂-GO] film.

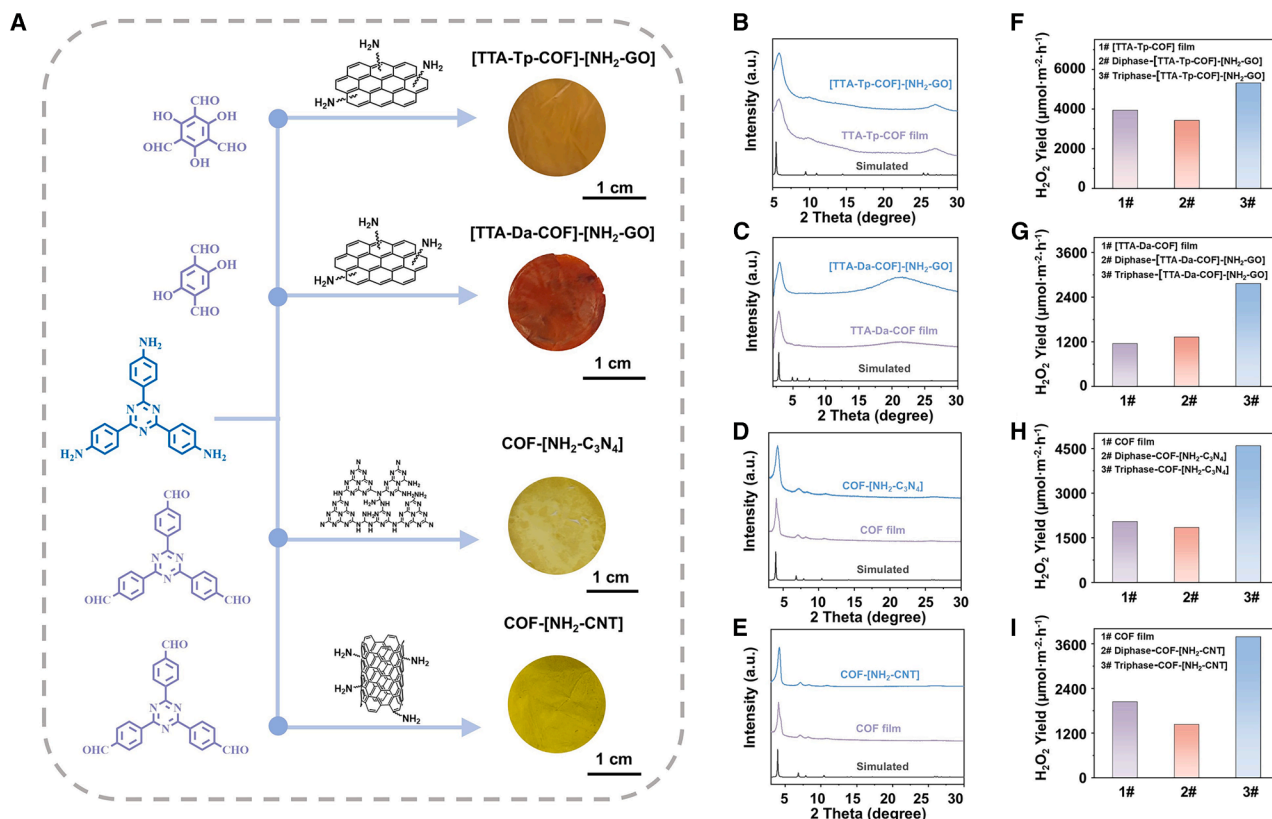
the concentration of H₂O₂ produced in the triphase system reached a maximum of 3.00 mM, considerably higher than that in the diphasic system (0.27 mM). These results demonstrated that the rapid transport of O₂ in the triphase configuration was a key factor for enhancing the photocatalytic H₂O₂ production rate. In addition, NH₂-GO was exposed in aqueous solution, and the H₂O₂ yield of the COF-[NH₂-GO] film was 3,145.8 μmol m⁻² g⁻¹ (Figure S39). However, when NH₂-GO was exposed in air, the H₂O₂ yield decreased to 1917.93 μmol m⁻² g⁻¹. It was confirmed that the orientation mode of the COF-[NH₂-GO] film influenced the photocatalytic performance.

To evaluate its practical potential, a film-based prototypical device was designed and fabricated for the sterilization of *E. coli* (Figure 4G). In this setup, pure water was pumped into a module packed with the COF-[NH₂-GO] film catalyst under visible light irradiation, and the concentration of as-synthesized H₂O₂ solution was about 3 mM. The as-collected solution was

subsequently used for antibacterial experiments against *E. coli*. As shown in Figure 4H, the growth of *E. coli* was effectively inhibited when the bacterial culture was co-cultured with a H₂O₂ solution obtained from our photocatalytic system. We further performed numerical simulations to model the dynamics of fluid flow and H₂O₂ transport within the film-based device (Figure S40). These results demonstrated not only the successful photocatalytic production of H₂O₂ in this device but also its strong antibacterial efficacy, highlighting the potential of the system for practical water disinfection applications.

The general strategy for synthesizing COF-based Janus films

To demonstrate the universality of the synthesizing approach for COF-based Janus film, we systematically prepared a series of composite films by varying both COFs or functional nanomaterials. As shown in Figure 5A, three different COFs (including TTA-Tp-COF, TTA-Da-COF, and the original COF) and two types of



functional nanomaterials, namely aminated graphitic carbon nitride (NH₂-C₃N₄) and aminated carbon nanotubes (NH₂-CNT), were employed. As shown in Figures 5A and S41, the corresponding optical photographs showed that all the prepared COF and COF-based composite films were uniform and intact. Contact angle measurements revealed the asymmetry wettability of the two sides for these films. After integrating with functional nanomaterials, the hydrophilicities of top surfaces (in H₂O phase) for these composite films increased to some extent (Figures S42–S47). As shown in Figures S48–S53, the SEM images of these films showed the different morphologies in their top and bottom surfaces, and the functional nanomaterials were uniformly dispersed on the film top surface. The above results show that all combinations successfully yielded COF-based Janus films with well-defined heterostructures. The XRD patterns of four composite films all exhibited diffraction peaks corresponding to COFs, confirming the crystalline structures of COFs in these composite films (Figures 5B–5E). In addition, all composite films showed visible light absorption relative to the pristine COF films (Figures S53–S57). The band gap widths and the VB positions were determined by Tauc diagrams and Mott Schottky measurements, respectively (Figures S58 and S59). Photocatalytic H₂O₂ production tests demonstrated that the self-floating COF-based Janus films in the

triphase system achieved the highest H₂O₂ yields (Figures 5F–5I; Tables S12–S15), outperforming both the COF films and the corresponding diphasic systems. Among them, the [TTA-Tp-COF]-[NH₂-GO] film delivered the highest H₂O₂ production rate, reaching up to 5,306.6 μmol·m⁻²·h⁻¹ (Figure 5F; Tables S10). These results highlighted the consistent synergistic enhancement effect between the COF and functional nanomaterials, as well as the advantage of the self-floating triphase structure. Furthermore, as for all composite films, the smaller half arc in EIS Nyquist plots and higher photocurrent density demonstrated better separation and transfer of photogenerated carriers (Figures S60–S67). In summary, despite variations in either organic or inorganic constituents, all four materials maintained structural integrity while exhibiting consistently improved light absorption, photocatalytic H₂O₂ production, and charge separation behavior. These findings collectively confirmed the universality and superiority of the one-step interfacial polymerization approach for fabricating functional self-floating COF-based Janus film for photocatalytic applications.

Conclusions

In this work, we employed a facile one-step interfacial polymerization strategy to fabricate a self-floating COF-[NH₂-GO] Janus film for H₂O₂ photosynthesis. The introduced graphene in this

unique heterostructure effectively promoted the separation and transfer of photogenerated carriers, while the self-floating film photocatalysts stabilized the gas-liquid-solid triphase interface to significantly enhance O₂ mass transfer. These factors collectively boosted the photocatalytic production of H₂O₂. Moreover, the practical applicability of the film was demonstrated by designing a film-based device capable of continuously generating H₂O₂ for the sterilization of *E. coli*. Importantly, the generality of interfacial assembly strategy for fabricating COF-based Janus films was verified through its successful extension to other COF frameworks and functional nanomaterials (i.e., C₃N₄ and CNT). This work not only presents a prospect for triphase artificial photosynthesis (e.g., H₂O₂ synthesis, CO₂ reduction, nitrogen fixation, and organic pollutant degradation) but also offers an insightful paradigm for designing advanced functional composite films.

METHODS

Materials

2,4,6-tris(4-formylphenyl)-1,3,5-triazine (TTB, C₂₄H₁₅N₃O₃, 97%), 2,4,6-tris(4-aminophenyl)-1,3,5-triazine (TTA, C₂₁H₁₈N₆, 98%), and 2,5-dihydroxyterephthalaldehyde (Da, C₈H₆O₄, 98%) were purchased from Shanghai Haohong Bio-pharmaceutical Technology Co., Ltd. Potassium iodide (KI, 99%) was obtained from Bide Pharmatech Co., Ltd. 2,4,6-Triformylphloroglucinol (Tp, C₉H₆O₆, 99%) was provided by Yanshen Technology Co., Ltd. Potassium hydrogen phthalate (C₈H₅KO₄, 99%) and urea (CH₄N₂O) were purchased from Aladdin. Acetic acid (C₂H₄O₂), dichloromethane (CH₂Cl₂), N, N-dimethylformamide (C₃H₇NO), ethanol (C₂H₅OH), tetrahydrofuran (C₄H₈O), acetone (C₃H₆O, ≥99.5%) and methanol (CH₄O) were analytical grade, provided by Tianjin Damao Chemical Reagent. 3-Aminopropyltriethoxysilane (APTS, C₉H₂₃NO₃Si, ≥98%), 1,2-dichlorobenzene (*o*-DCB, C₆H₄Cl₂, ≥99%), and 1-butanol (*n*-BuOH, C₄H₁₀O, ≥99.5%) were purchased from Aladdin. All chemicals were commercially available and used without further purification.

Characterization

The powder XRD pattern was collected on the smart X-ray diffractometer (SmartLab 9 kW, Rigaku, Japan) equipped with Cu K α radiation ($\lambda = 1.54 \text{ \AA}$). FTIR spectroscopy and *in situ* FTIR spectra were recorded on Nicolet iS50 IR spectrometers (Thermo Scientific, USA), and samples were tableted with KBr as support. UV-Vis absorption spectra were recorded on a Hitachi U-3900 UV/VIS spectrophotometer (Hitachi, Japan). SEM was performed on a Quanta250FEG scanning electron microscope (FEI, America). HRTEM images and element mapping were collected on Talos F200 \times transmission electron microscopes (FEI, America). UV-Vis DRS measurements were obtained with a Lambda 750 UV/VIS/NIR spectrometer (PerkinElmer, America). EPR spectroscopy was performed on an ESR5000 X-band spectrometer (Bruker Magnetech, Germany). The PL spectra were collected using a F-7000 fluorescence spectrophotometer (Hitachi, Japan). Time-resolved confocal fluorescence microscopy was performed with a Microtime 200 system (PicoQuant, Germany). Contact angle

measurements were performed with an SZ-CAMC33 contact angle meter (SZ, China). The isotope of ¹⁸O for O₂ was analyzed using an HPR-20 QIC mass spectrometry system (Hidden Analytical, UK).

Synthesis of COF powder

COF powder synthesis followed a procedure reported in the literature.³¹ A mixture of TTA (23 mg, 0.085 mmol), TTB (30 mg, 0.085 mmol), *o*-DCB (0.75 mL), *n*-BuOH (0.75 mL), and 6 M aqueous acetic acid (0.1 mL) was added into a Pyrex tube. Then, the tube was sonicated for 15 min, followed by flash freezing at 77 K in a liquid N₂ bath, which was further evacuated to an internal pressure of 50 mTorr and sealed off. The resulting mixture was heated at 120°C for 72 h. After cooling to room temperature, the yellow solid formed was filtered and washed several times with acetone/methanol. Finally, the product was dried at 80°C under vacuum.

Synthesis of COF film

COF film was synthesized via interfacial polymerization at the CH₂Cl₂/water interface according to the following optimized procedure. First, 0.006 mmol of TTB dissolved in 3 mL of CH₂Cl₂ was added to a 20-mL glass bottle. Subsequently, 6 mL deionized water was carefully layered on the CH₂Cl₂ phase to form a two-phase interface. Finally, 0.006 mmol of TTA dissolved in 3 mL of acetic acid (3 M) was slowly added to the topmost layer of the aqueous phase. The reaction system was kept undisturbed at room temperature for 7 days. After the reaction, the film formed at the interface was collected, washed with CH₂Cl₂, N,N-Dimethylformamide (DMF), and ethanol, and dried at room temperature.

Synthesis of NH₂-GO

NH₂-GO was synthesized according to a previous report.⁵⁷ 0.1 g of GO was dispersed in 35 mL of deionized water and sonicated for 2 h to obtain a homogeneous dispersion. Then, 0.2 mL APTS was added to the mixture, with stirring for 4 h. After the reaction, the product (NH₂-GO) was collected by centrifugation and thoroughly washed using deionized water for 5 times to remove residual reactants.

Synthesis of COF-[NH₂-GO]

COF-[NH₂-GO] was synthesized via interfacial polymerization at the CH₂Cl₂/water interface according to the following optimized procedure. First, 0.006 mmol of TTB dissolved in 3 mL of CH₂Cl₂ was added to a 20-mL glass bottle. Subsequently, 0.9 mg of NH₂-GO dissolved in 6 mL of deionized water was carefully layered on the CH₂Cl₂ phase to form a two-phase interface. Finally, 0.006 mmol of TTA dissolved in 3 mL of acetic acid (3 M) was slowly added to the topmost layer of the aqueous phase. The reaction system was kept undisturbed at room temperature for 7 days. After the reaction, the film formed at the interface was collected, washed with CH₂Cl₂, DMF, and ethanol, and dried at room temperature.

Synthesis of NH₂-C₃N₄

NH₂-C₃N₄ was synthesized according to a previous report.⁵⁸ N[(CH₂)₂NHCH₂(*m*-C₆H₄)CH₂NH(CH₂)₂]₃N (12.5 μ mol) and urea

(5.0 g) were dissolved in deionized water. The solution was concentrated by heating, and the remaining water was removed by rotary evaporation to obtain a uniform powder. The powder was then transferred to a crucible and pyrolyzed for 2 h under ambient air at 600°C (heating rate: 5°C/min). The product obtained was washed with deionized water and ethanol, and then dried under vacuum at 60°C to obtain light brown powder NH₂-C₃N₄.

Synthesis of COF-[NH₂-C₃N₄]

COF-[NH₂-C₃N₄] was synthesized via interfacial polymerization at the CH₂Cl₂/water interface according to the following optimized procedure. First, 0.006 mmol of TTB dissolved in 3 mL of CH₂Cl₂ was added to a 20-mL glass bottle. Subsequently, 0.9 mg of NH₂-C₃N₄ dissolved in 6 mL deionized water was carefully layered on the CH₂Cl₂ phase to form a two-phase interface. Finally, 0.006 mmol of TTA dissolved in 3 mL of acetic acid (3 M) was slowly added to the topmost layer of the aqueous phase. The reaction system was kept undisturbed at room temperature for 7 days. After the reaction, the film formed at the interface was collected, washed with CH₂Cl₂, DMF, and ethanol, and dried at room temperature.

Synthesis of NH₂-CNT

NH₂-CNT was synthesized from a previous report.⁵⁹ COOH-CNT (0.05 g) was dispersed in 50 mL ethanol and ultrasonicated for 30 min. Subsequently, APTS was added to the mixture, then stirred at 70°C for 4 h. After the reaction, the product was washed with water and acetone.

Synthesis of COF-[NH₂-CNT]

COF-[NH₂-CNT] was synthesized via interfacial polymerization at the CH₂Cl₂/water interface according to the following optimized procedure. First, 0.006 mmol of TTB dissolved in 3 mL of CH₂Cl₂ was added to a 20-mL glass bottle. Subsequently, 0.9 mg of NH₂-CNT dissolved in 6 mL deionized water was carefully layered on the CH₂Cl₂ phase to form a two-phase interface. Finally, 0.006 mmol of TTA dissolved in 3 mL of acetic acid (3 M) was slowly added to the topmost layer of the aqueous phase. The reaction system was kept undisturbed at room temperature for 7 days. After the reaction, the film formed at the interface was collected, washed with CH₂Cl₂, DMF, and ethanol, and dried at room temperature.

Synthesis of [TTA-Da-COF]-[NH₂-GO]

[TTA-Da-COF]-[NH₂-GO] was synthesized via interfacial polymerization at the CH₂Cl₂/water interface according to the following optimized procedure. First, 0.006 mmol of Da dissolved in 3 mL CH₂Cl₂ was added to a 20-mL glass bottle. Subsequently, 0.9 mg NH₂-GO dissolved in 6 mL deionized water was carefully layered on the CH₂Cl₂ phase to form a two-phase interface. Finally, 0.006 mmol of TTA dissolved in 3 mL of acetic acid (3 M) was slowly added to the topmost layer of the aqueous phase. The reaction system was kept undisturbed at room temperature for 7 days. After the reaction, the film formed at the interface was collected, washed with CH₂Cl₂, DMF, and ethanol, and dried at room temperature.

Synthesis of [TTA-Tp-COF]-[NH₂-GO]

[TTA-Tp-COF]-[NH₂-GO] was synthesized via interfacial polymerization at the CH₂Cl₂/water interface according to the following optimized procedure. First, 0.006 mmol of Tp dissolved in 3 mL CH₂Cl₂ was added to a 20-mL glass bottle. Subsequently, 0.9 mg of NH₂-GO dissolved in 6 mL deionized water was carefully layered on the CH₂Cl₂ phase to form a two-phase interface. Finally, 0.006 mmol of TTA dissolved in 3 mL of acetic acid (3 M) was slowly added to the topmost layer of the aqueous phase. The reaction system was kept undisturbed at room temperature for 7 days. After the reaction, the film formed at the interface was collected, washed with CH₂Cl₂, DMF, and ethanol, and dried at room temperature.

Photocatalytic test

In a typical experiment, 20 mL deionized water was added to a beaker (*r* = 2 cm), and the film was placed at the air-water interface. Then, the reaction system was irradiated using a 300 W xenon lamp with a UV cut-off filter ($\lambda > 420$ nm) for 1 h.

H₂O₂ detection methods

The concentration of H₂O₂ was measured by iodimetry. 1 mL of 0.4 M KI aqueous solution and 1 mL of 0.1 M C₈H₅KO₄ aqueous solution were added to 1 mL of the reaction solution. The mixture was mixed thoroughly and allowed to stand for 30 min. H₂O₂ reacted with iodine ions (I⁻) to form triiodide ions (I₃⁻) under acidic conditions, which led to strong absorption near 350 nm. The absorbance at 350 nm was measured by UV-vis spectrophotometer, and the quantity of H₂O₂ was calculated.

COF-[NH₂-GO] film-based device for H₂O₂ photogeneration

To demonstrate practical application, a film-based device was fabricated by integrating the COF-[NH₂-GO] film into a quartz cell equipped with inlet and outlet ports. The setup was irradiated from the top using an Xe lamp. The H₂O₂ solution produced during photocatalysis was collected from the outlet for subsequent antibacterial evaluation.

Sterilization of *E. coli*

The antibacterial efficacy of the obtained H₂O₂ solution was evaluated using the plate colony counting method, with *E. coli*. Initially, a single colony was inoculated into fresh LB medium and incubated on a shaker at 37°C and 200 rpm for 12 h. The initial culture was then diluted 100-fold and further cultured under the same conditions for 3 h to obtain a secondary culture in the logarithmic growth phase (approximately 10⁶ CFU/mL). Subsequently, 100 μ L of this bacterial suspension was mixed with 500 μ L of the obtained H₂O₂ solution and 50 μ L FeSO₄·7H₂O and incubated for 3 h. The mixtures were then plated, and after incubation at 37°C for 24 h, the colonies were photographed.

Theoretical calculation

All DFT calculations were performed using the Gaussian 16 package.⁶⁰ Geometry optimizations were carried out at the B3LYP/def2-SVP level of theory⁶¹ with Grimme's D3 dispersion correction.^{62–64} The SMD solvation model⁶⁵ employing water as

the solvent was applied during all geometry optimizations. Quadratic convergence criteria were used for self-consistent field (SCF) calculations,⁶⁶ and the XQC algorithm was invoked when the standard SCF procedure failed to converge. Vibrational frequency analyses were performed under the same conditions as those used for geometry optimization. Single-point energy calculations were subsequently conducted at the M06/def2-TZVP level to obtain more accurate electronic energies.⁶⁷ All optimized structures were visualized using CYLview.⁶⁸

Numerical simulations via FEM and PINNs

The governing equations for mass transport read:

$$D_i \nabla^2 c_i + S_i = 0,$$

where i stands for different species, c for molar concentration, D for diffusivity, S for the source term (or more specifically, consumption/production rate per unit volume). For both the diphase and triphase systems, the thickness of the diffusion layer was set as 5 mm, and the thickness of the COF film located in between was set to be $h = 5 \mu\text{m}$. At the top boundary for the triphase system, the O₂ partial pressure was assumed to be $p = 0.21 \text{ atm}$, and the corresponding concentration could be calculated according to the ideal gas law: $c = p/RT$; for the boundaries in aqueous phase, the O₂ concentration was assumed to be $0.27 \text{ mol}\cdot\text{m}^{-3}$ (according to Henry law). The diffusivities were set to be $2.1 \times 10^{-9} \text{ m}^2 \text{ s}^{-1}$ and $1.5 \times 10^{-9} \text{ m}^2 \text{ s}^{-1}$ for O₂ and H₂O₂, respectively, in the aqueous/solid phase and $2.1 \times 10^{-5} \text{ m}^2 \text{ s}^{-1}$ for O₂ in the gas phase. The source terms S for the liquid and gas phases were set as 0 and calculated according to $S_i = J_i/h$ for the solid phase, where J_i is the experimentally determined consumption/production rate per unit area (unit, $\text{mol m}^{-2} \text{ s}^{-1}$). At the air-solid interface, the Henry constant of 0.032 was set for O₂. At the interfaces between the liquid/solid/gas phases, proper boundary conditions regarding the continuities of chemical potentials and transport fluxes were prescribed.

For the FEM, a uniform mesh of 5 mm and a quadratic shape function were used. For the PINNs, three networks (each for a liquid, solid, or gas phase) with two hidden layers (each with 32 neurons) were used; the activation functions were set as hyperbolic tangent for the hidden layers and linear for the output layer. Wherever a Dirichlet boundary condition was prescribed, a hard constraint was implemented. Mean square error losses were used to monitor the numerical residuals for the governing equation as well as the continuities of chemical potentials and transport fluxes at the liquid/solid/gas interfaces; the weights for different losses were dynamically adjusted during training. For each network, an Adam (Adaptive Momentum) optimizer was employed with a learning rate of 0.001.

RESOURCE AVAILABILITY

Lead contact

Further information and requests for resources and reagents should be directed to and will be fulfilled by the lead contact, Kuo Yuan (yuankuo@email.tjut.edu.cn).

Materials availability

This study did not generate new, unique reagents.

Data and code availability

The data and code associated with the study have not been deposited in a public repository but are available from the lead contact upon reasonable request.

ACKNOWLEDGMENTS

This work was supported by the National Natural Science Foundation of China (22575173, 22205162, 22571228, 22531007, 22271218, 22201209, and 22571227) and the Natural Science Foundation of Tianjin (nos. 24JCZDJC00220, 25JCQNJC00150, and 25JCQNJC00480).

AUTHOR CONTRIBUTIONS

Y.L. conducted the experiments and prepared the first draft of the manuscript; Y.Z. revised and polished the manuscript, characterized films, and analyzed experimental data; W.S. conducted the theoretical calculation; C.Z. conducted numerical simulations and physics-informed neural networks; Q.Y. revised the manuscript; Z.L. revised the supporting information; F.Y. and J.G. conducted the sterilization of *E. coli*; Q.Y. provided help in numerical simulations; W.L. and M.W. helped determine the crystal structures of COFs; X.Z. supported spectrum characterization; W.H. revised the film characterization part in the manuscript; K.Y. provided the idea of Janus film photocatalyst, finished the preparation of manuscript and supporting information, and supervised this work; D.Z. provided suggestion for photocatalysis; and T.L. revised the manuscript.

DECLARATION OF INTERESTS

The authors declare no competing interests.

SUPPLEMENTAL INFORMATION

Supplemental information can be found online at <https://doi.org/10.1016/j.matt.2026.102679>.

Received: December 21, 2025

Revised: January 27, 2026

Accepted: January 29, 2026

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