

In Situ Coating CsPbBr₃ Nanocrystals with Graphdiyne to Boost the Activity and Stability of Photocatalytic CO₂ Reduction

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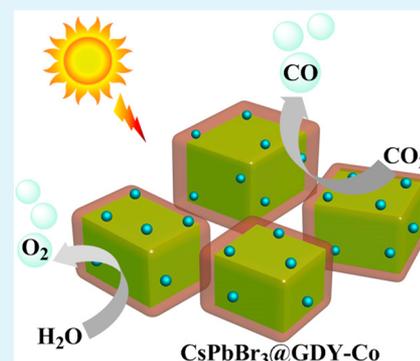
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ABSTRACT: The instability and low inferior catalytic activity of metal-halide perovskite nanocrystals are crucial issues for promoting their practical application in the photocatalytic field. Herein, we in situ coat a thin graphdiyne (GDY) layer on CsPbBr₃ nanocrystals based on a facile microwave synthesis method, and employ it as a photocatalyst for CO₂ reduction. Under the protection of GDY, the CsPbBr₃-based photocatalyst delivers significantly improved stability in a photocatalytic system containing water concomitant with enhanced CO₂ uptake capacity. The favorable energy offset and close contact between CsPbBr₃ and GDY trigger swift photogenerated electron transfer from CsPbBr₃ to doping metal sites in GDY, boosting a remarkable improvement in the photocatalytic performance for CO₂ reduction. Without adding traditional sacrificial reductants, the cobalt-doped photocatalyst achieves a high yield of 27.7 μmol g⁻¹ h⁻¹ for photocatalytic CO₂ conversion to CO based on water as a desirable electron source, which is about 8 times higher than that of pristine CsPbBr₃ nanocrystals.

KEYWORDS: CO₂ reduction, photocatalysis, perovskite nanocrystal, in situ coating, graphdiyne



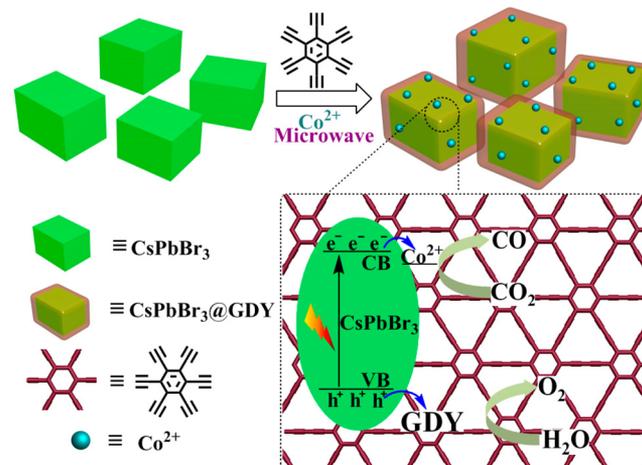
1. INTRODUCTION

The photocatalytic reduction of CO₂ to chemical feedstocks or fuels, using cost-effective photosynthesis systems, is an ideal strategy to reduce both the consumption of fossil fuels and the intensification of the greenhouse effect.^{1–4} In this artificial photosynthesis technology, most of the designed systems separate the two halves of the reduction and oxidation reactions using expensive sacrificial electron donors or acceptors to close the catalytic cycle.^{5,6} However, the reduction of CO₂ simultaneously using water as an electron donor is ultimately desired for practical application. In this regard, metal-halide perovskite (MHP) nanocrystals, which possess fascinating light-harvesting performance, excellent optoelectronic properties, and are cost-effective,^{7–9} have recently been utilized to achieve photocatalytic reduction of CO₂ concurrent with desirable water oxidation,^{10–15} demonstrating great potential in the field of practical artificial photosynthesis.^{16–19} However, there are two problems in MHP nanocrystal based photocatalytic systems that need to be solved urgently for its successful application in artificial photosynthesis. First, the ionic nature of MHP nanocrystals severely reduces their stability in reaction systems containing plenty of water.^{20–22} Second, the lack of effective catalytic sites and inferior charge separation limit them for high photocatalytic performance.^{13,23–28} To date, these obstacles still require facile and effective solutions.

In order to settle these open questions, in this contribution, we in situ coat a thin graphdiyne (GDY) layer, a new member of the carbon family,²⁹ onto the surface of CsPbBr₃

nanocrystals (coded as CsPbBr₃@GDY) based on a facile microwave synthesis method (Scheme 1),³⁰ and employ it as a

Scheme 1. Illustration of the Synthesis of Cobalt Doped CsPbBr₃@GDY and Its Photoredox Reactions



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compatible platform for the decoration of metal ions as active sites for photocatalytic CO₂ reduction. Our strategy is mainly based on the following important considerations: (1) Coating the all-carbon material GDY on the surface of nanocrystals can improve the structural and interfacial stabilities of the nanocrystals.³¹ (2) The sp-hybridized carbon atoms and triangular cavities in the GDY facilitate the metal atom doping to increase the active sites for photocatalytic reaction.^{32,33} (3) The favorable energy offset and close contact between CsPbBr₃ and GDY can trigger swift photogenerated carrier transfer between CsPbBr₃ and GDY, thus boosting the improvement of the photoinduced charge separation efficiency of CsPbBr₃ and the catalytic activity of the composite photocatalyst. As expected, the CsPbBr₃@GDY composite photocatalysts deliver significantly improved stability in a photocatalytic system containing water. Using water as an electron source, a high yield of 27.7 μmol g⁻¹ h⁻¹ for visible-light-driven CO₂ conversion to CO has been achieved by doping cobalt ions.

2. RESULTS AND DISCUSSION

2.1. Synthesis and Characterization of CsPbBr₃@GDY.

The CsPbBr₃@GDY nanocomposite was prepared with a two-step synthetic method, and the detailed procedures are described in the [Experimental Section](#). Briefly, CsPbBr₃ nanocrystals were first prepared via the solution phase synthesis strategy according to a previous report (see details in the [Supporting Information, SI](#)).¹⁵ Then the as-prepared CsPbBr₃ nanocrystals, acting as a growth substrate, were mixed with different amounts of hexaethynylbenzene monomer in a microwave reactor and heated for 10 min ([Scheme 1](#)). Depending on the amount of hexaethynylbenzene monomer, the obtained samples were labeled as CsPbBr₃@GDY_{0.1}, CsPbBr₃@GDY_{0.3}, and CsPbBr₃@GDY_{0.5}, respectively. For comparison, pristine GDY was also prepared under the same conditions in the absence of CsPbBr₃ nanocrystals ([Figure S1](#)).³⁰ Transmission electron microscopy (TEM) measurements ([Figure 1a and 1b](#)) show that both CsPbBr₃ and

CsPbBr₃@GDY present as cubic nanostructures with a particle size of 10–20 nm. For the CsPbBr₃@GDY_{0.3} sample, the GDY layer wrapped tightly around the surface of the CsPbBr₃ nanocrystal and formed a shell structure with a thickness of 1.9 nm (inset of [Figure 1b](#)). In contrast, the HRTEM image of pure CsPbBr₃ ([Figure S2](#)) shows relatively narrower and more ambiguous edges around the CsPbBr₃ nanocrystal, which may result from the adhesion of oleic acid and oleylamine residuals on the surface of CsPbBr₃ ([Figure S3](#)). Powder X-ray diffraction (XRD) measurements ([Figure 1c](#)) show that CsPbBr₃@GDY_{0.1}, CsPbBr₃@GDY_{0.3}, and CsPbBr₃@GDY_{0.5} have the same crystal forms as individual CsPbBr₃, characteristic of a cubic phase (JCPDS 00-054-0752), indicating that the crystal structure of CsPbBr₃ is maintained during the process of in situ coating. The Raman spectra of CsPbBr₃@GDY_{0.1}, CsPbBr₃@GDY_{0.3}, and CsPbBr₃@GDY_{0.5} ([Figure S4](#)) show two characteristic peaks at 1930.6 and 2102.8 cm⁻¹, which can be ascribed to the typical signals of acetylene bonds and conjugated diyne chains, evidencing the successful synthesis of GDY on CsPbBr₃ nanocrystals.^{29,34} Furthermore, X-ray photoelectron spectroscopy (XPS) measurements were conducted to illustrate the chemical bonds in CsPbBr₃@GDY_{0.3}. The C 1s spectrum of CsPbBr₃@GDY_{0.3} ([Figure 1d](#)) shows that the carbon species of CsPbBr₃@GDY_{0.3} can be deconvoluted into six peaks, including C–C(sp³), C=C(sp²), C≡C(sp), C–N, C–O, and C=O.^{35,36} The calculated integral ratio of sp- and sp²-hybridized carbon is close to 2, further confirming the successful coating of GDY onto CsPbBr₃ nanocrystals.^{29,30}

2.2. Photocatalytic Reduction of CO₂. The coating effect of GDY on the photocatalytic performance of CsPbBr₃ for CO₂ reduction was assessed by adjusting the amount of introduced hexaethynylbenzene monomer during the microwave synthesis process. Specifically, the photoreactions were carried out in a CO₂-saturated solution of acetonitrile with H₂O as the electron donor. A xenon lamp equipped with a 400 nm filter was used as the illuminant, with a light intensity of 100 mW cm⁻². The results of gas chromatographic measurements and ¹³C nuclear magnetic resonance spectroscopy ([Figure 2a, Figure S5 and Table S1](#)) imply that the major product of CO₂ reduction is CO for all the evaluated photocatalysts, and the in situ coating of GDY can significantly enhance the photocatalytic activity of CsPbBr₃. Herein, the adverse production of CH₄ may result from the competing desorption of CO, which could act as an intermediate for the conversion of CH₄.³⁷ Besides, increasing the content of GDY results in a volcanic tendency of CO production. The evolution of CO reaches an optimal value of 10.2 μmol g⁻¹ h⁻¹ when CsPbBr₃@GDY_{0.3} is used as photocatalyst, 3.1 times that of pure CsPbBr₃ (3.3 μmol g⁻¹ h⁻¹). The activity degradation of CsPbBr₃@GDY_{0.5} may originate from the significant opacity and severe light scattering caused by excessive GDY, suggesting that an appropriate amount of GDY coating is critical to improve the photocatalytic performance of CsPbBr₃. It is worth noting that the amount of surface oleic acid and oleylamine residuals could also effect the composition between CsPbBr₃ and GDY. Sufficient washing times of CsPbBr₃ are significant to balance the carrier transport between CsPbBr₃ and GDY as well as the stability of the CsPbBr₃@GDY_{0.3} composite ([Figure S6](#)). For comparison, the photocatalytic activity of the physical mixture of CsPbBr₃ and GDY was also tested (2.1 μmol g⁻¹ h⁻¹); however, no enhancement was observed compared with pure CsPbBr₃, indicating the

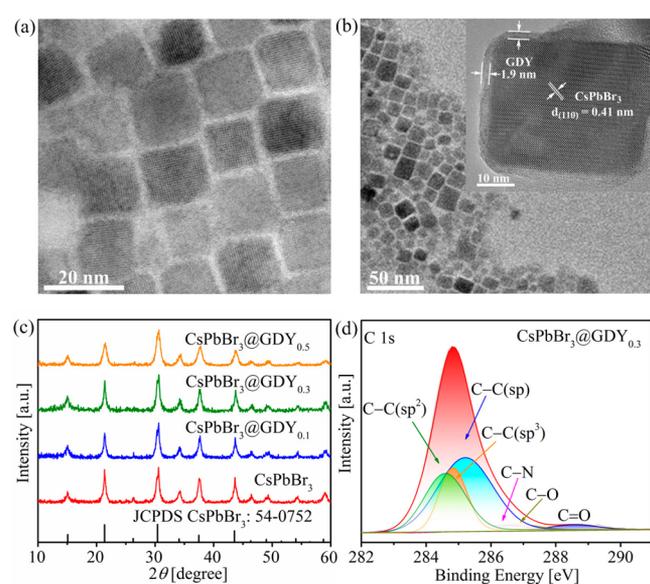


Figure 1. TEM images of (a) CsPbBr₃ and (b) CsPbBr₃@GDY_{0.3} (inset: HRTEM image). (c) XRD patterns of CsPbBr₃ and CsPbBr₃@GDY_{0.1–0.5}. (d) XPS spectra of C 1s for CsPbBr₃@GDY_{0.3}.

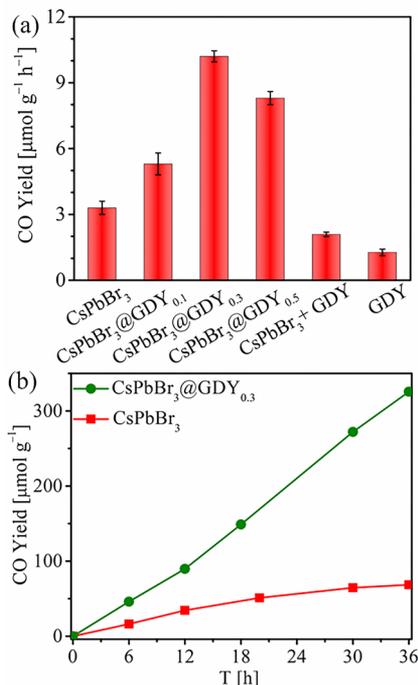


Figure 2. (a) The yields of CO productions from photocatalytic CO₂ reduction using CsPbBr₃, CsPbBr₃@GDY_{0.1–0.5}, a mixture of CsPbBr₃ and GDY, and GDY as photocatalysts. (b) The tracking of CO production over time.

remarkable benefit of close contact between CsPbBr₃ and GDY for the separation of photogenerated carriers. For pristine GDY, which displays a natural band gap and has been applied as a photoactive material in many reports,^{38–40} trace amounts of CO were detected (1.2 μmol g⁻¹ h⁻¹), suggesting that GDY itself was almost inactive for photoreduction of CO₂ in this system. Furthermore, the impact of GDY coating on the stability of CsPbBr₃ was also investigated in the photocatalytic experiment. As shown in Figure 2b, there is no observable decrease of CO evolution for CsPbBr₃@GDY_{0.3} after 36 h of photocatalytic reaction. In contrast, the photocatalytic activity of pure CsPbBr₃ shows an obvious stagnation after 12 h illumination. In addition, the XRD and Raman spectra measurements (Figures S7 and S8) indicated that there are no significant changes in crystal structure and Raman signals for CsPbBr₃@GDY_{0.3} before and after photoreaction, demonstrating the stability of CsPbBr₃@GDY_{0.3} during the photoreaction. While for CsPbBr₃, some new peaks related to the crystalline phase of CsPb₂Br₅ are observed after photoreaction (Figure S9), indicating that the stability of CsPbBr₃ is relatively weak compared with CsPbBr₃@GDY. These comparative results suggest that the coating of GDY can improve the activity and stability of CsPbBr₃ nanocrystals photocatalysts concurrently.

To verify the origin of CO product, a ¹³CO₂ isotope labeling experiment (Figure S10) was conducted. The gas chromatography–mass spectrometry (GC–MS) measurement shows that an *m/z* = 29 corresponding to ¹³CO was clearly detected, indicating that CO is the product of the CO₂ reduction. Additionally, increasing O₂ was also detected in the gas system besides the CO product during photoreactions. To verify that O₂ originates from the photo-oxidation of H₂O, isotopic experiment with H₂¹⁸O was conducted. The signal at *m/z* = 36

corresponding to ¹⁸O₂ was detected by GC–MS measurement (Figure S11), demonstrating that H₂O is the electron donor for the annihilation of photogenerated holes in CsPbBr₃@GDY_{0.3} catalyst.

2.3. Coating Effect of GDY. To better understand the origin of the photocatalytic enhancement induced by the in situ coating of GDY on the CsPbBr₃ surface, the electronic band structures of GDY and CsPbBr₃ were first measured, which have a crucial influence on the interfacial charge transfer between GDY and CsPbBr₃. As shown in Figure S12, the band gap of CsPbBr₃ can be determined via the UV–vis spectrum of CsPbBr₃, which is 2.37 eV. The conduction band (CB) potentials of CsPbBr₃ and GDY derived from the Mott–Schottky plots (Figure S13) are –1.07 and –0.98 V versus the normal hydrogen electrode (NHE), respectively. Furthermore, the valence band (VB) spectra of XPS suggest that the VB potential of GDY is located 0.3 eV above CsPbBr₃ (Figure S14). Accordingly, the relative energy-band positions of CsPbBr₃ and GDY can be diagrammed in Figure 3a. It can

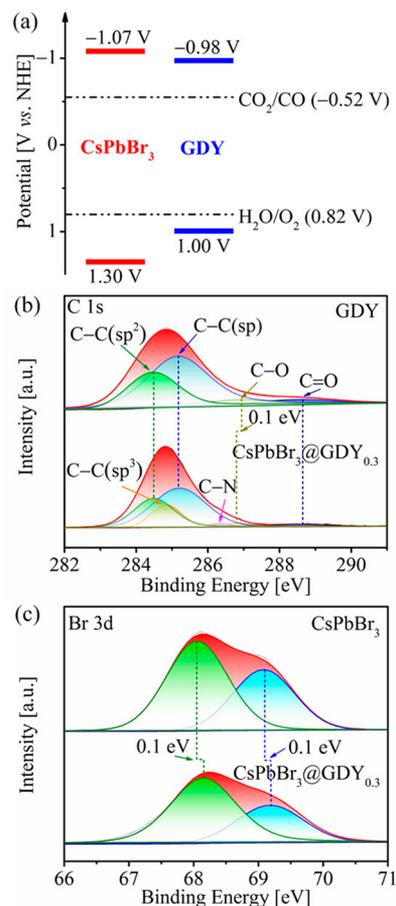


Figure 3. (a) Energy band structures of CsPbBr₃ and GDY. (b) XPS spectra of C 1s for GDY and CsPbBr₃@GDY_{0.3}. (c) XPS spectra of Br 3d for CsPbBr₃ and CsPbBr₃@GDY_{0.3}.

be noted that both of the photogenerated electrons in the CBs of CsPbBr₃ and GDY are energetically competent to trigger the reduction of CO₂ to CO; however, owing to the poor separation efficiency of photogenerated carriers, pristine GDY shows almost no detectable photoactivity toward CO₂ reduction (Figure 2a) or other reaction.⁴¹ Besides, as shown in Figure 3a, the photogenerated holes of CsPbBr₃ can transfer

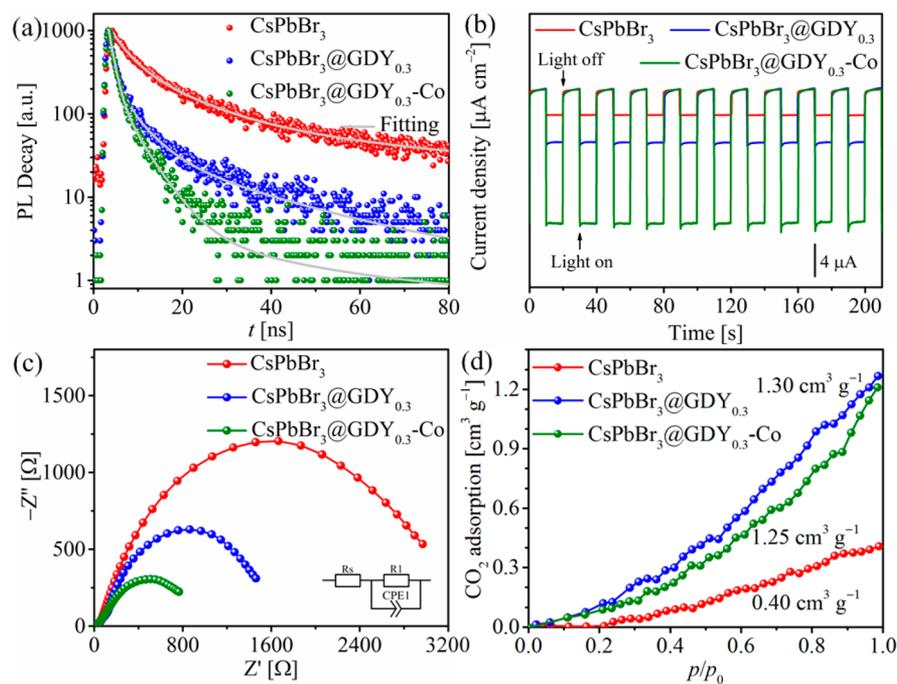


Figure 4. (a) PL decay traces, (b) $I-t$ curves, and (c) EIS spectra of CsPbBr₃, CsPbBr₃@GDY_{0.3}, and CsPbBr₃@GDY_{0.3}-Co under light irradiation. (d) The adsorption curves of CO₂ for CsPbBr₃, CsPbBr₃@GDY_{0.3}, and CsPbBr₃@GDY_{0.3}-Co.

to the VB of GDY, which could improve the separation efficiency of the photogenerated charges in CsPbBr₃.

XPS measurements were further carried out to analyze the interface interaction behaviors between CsPbBr₃ and GDY. As presented in Figure 3b, the subpeak corresponding to C–O in the C 1s peak of CsPbBr₃@GDY_{0.3} shows a perceptible shift to low energy (by ~ 0.1 eV) compared to that of pristine GDY. Meanwhile, there are distinct positive shifts of Br 3d_{3/2} and Br 3d_{5/2} peaks (~ 0.1 eV) in CsPbBr₃@GDY_{0.3} with respect to that in pristine CsPbBr₃ (Figure 3c). In addition, the binding energies of Cs 3d (Figure S15) and Pb 4f (Figure S16) in CsPbBr₃@GDY_{0.3} also feature perceivable positive shifts in comparison with those in pure CsPbBr₃ nanocrystals. These shifts of binding energies indicate the occurrence of strong interfacial electronic coupling between CsPbBr₃ and GDY. The orientation of binding energy suggests that the electrons in the surface of CsPbBr₃ tend to migrate to GDY, resulting in a built-in electric field at the interface of CsPbBr₃@GDY_{0.3}, pointing from CsPbBr₃ to GDY. This built-in electric field can facilitate the photogenerated holes transfer from CsPbBr₃ to GDY, suggesting that the photogenerated electrons and holes of CsPbBr₃ could be separated efficiently.

The time-correlated single photon counting (TCSPC) technique was further utilized to study the detailed dynamics of photogenerated charge transfer between CsPbBr₃ and GDY. As presented in Figure 4a, the time-correlated photoluminescence (PL) decay traces could be well fitted with a three-exponential function to determine the averaged lifetime of PL decays (Table S2). The PL decay trace of pure CsPbBr₃ (red symbols in Figure 4a) featuring an averaged lifetime of 28.10 ns could be ascribed to the deactivation of photogenerated excitons in the CsPbBr₃ nanocrystals through radiative and nonradiative modes. Coating GDY onto CsPbBr₃ results in a significantly accelerated PL decay (blue symbols). The corresponding averaged PL lifetime of CsPbBr₃@GDY_{0.3} is 9.48 ns (Table S2), which is noticeably shorter than that of

pristine CsPbBr₃ (28.10 ns, Table S2), demonstrating the occurrence of efficient carrier transfer between CsPbBr₃ and GDY, owing to their strong interfacial interaction and favorable energy-offset. In addition, photocurrent response and electrochemical impedance spectroscopy (EIS) experiments were conducted to illustrate the benefit of GDY on the charge separation/transfer between CsPbBr₃ and GDY. As shown in Figure 4b, the photocurrent density of CsPbBr₃@GDY_{0.3} is notably larger than that of CsPbBr₃, indicating a more efficient photogenerated carrier separation of CsPbBr₃@GDY_{0.3} in comparison with pristine CsPbBr₃. The EIS of CsPbBr₃@GDY_{0.3} with a smaller arc diameter (Figure 4c) further demonstrates a higher electron mobility of CsPbBr₃@GDY_{0.3} compared to pure CsPbBr₃. These results once again confirm the beneficial impact of GDY on the interfacial carrier separation/transfer.

Apart from the separation efficiency of photogenerated carriers, the concentration of CO₂ on a photocatalyst also has an important influence on the photocatalytic activity. Thereby, the influence of coating GDY on the CO₂ uptake ability was further investigated by recording the CO₂ adsorption curves of CsPbBr₃ and CsPbBr₃@GDY_{0.3}. As depicted in Figure 4d, CsPbBr₃@GDY_{0.3} has an adsorption capacity of 1.3 cm³ g⁻¹ toward CO₂, which is 3.2 times higher than that of CsPbBr₃, suggesting that the coating of GDY can enrich CO₂ to the surface of the catalyst. The remarkable improvement in CO₂ adsorption capacity may be ascribed to the unique structure of GDY's characteristic carbon–carbon triple bonds with delocalized electrons and abundant distributed pores, which has been demonstrated to be beneficial for the CO₂ adsorption.^{42,43} Overall, coating GDY onto the surface of CsPbBr₃ can bring forth an enhanced gathering ability toward CO₂ and improved separation efficiency of photogenerated carriers, which jointly contribute to the aforementioned higher photocatalytic activity of CsPbBr₃@GDY_{0.3} compared to pure CsPbBr₃.

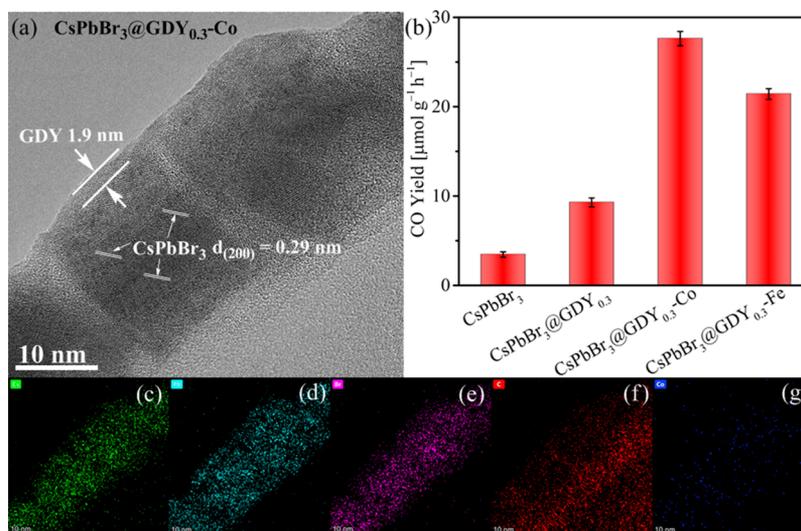


Figure 5. (a) HRTEM image for CsPbBr₃@GDY_{0.3}-Co; (b) photocatalytic CO production yields of CsPbBr₃, CsPbBr₃@GDY_{0.3}, CsPbBr₃@GDY_{0.3}-Co, and CsPbBr₃@GDY_{0.3}-Fe; and (c–g) elemental mapping images (Cs, Pb, Br, C, and Co) of CsPbBr₃@GDY_{0.3}-Co.

2.4. Doping of Co(II) on CsPbBr₃@GDY. As GDY contains abundant sp-hybridized carbons, which can serve as a favorable substrate for the deposition of metal species,^{44–46} Co(II) ions were chosen as catalytic sites to trap the photoexcited electrons and transfer them to reduce CO₂. The synthetic process for cobalt-doped CsPbBr₃@GDY_{0.3} (CsPbBr₃@GDY_{0.3}-Co) is similar to that of CsPbBr₃@GDY_{0.3}, except that CoBr₂ was added to the mixed solution of CsPbBr₃ nanocrystals and hexaethynylbenzene monomer during microwave synthesis (see Experimental Section). The decoration of Co(II) (Table S3, 1.56 wt %) on CsPbBr₃@GDY_{0.3} was characterized through XPS, TEM, and elemental mapping measurements. As shown in Figure S17, the binding energy of Co 2p_{1/2} at 780.9 eV demonstrates that the cobalt in CsPbBr₃@GDY_{0.3}-Co presents as Co²⁺.⁴⁷ Herein, the presence of Co is thought to be supported by GDY in a monodisperse form, as the doping of Co in CsPbBr₃@GDY_{0.3}-Co is through the post synthesis method, in which the prepared CsPbBr₃ nanocrystals were used as a substrate to support GDY and Co. Thus, it is difficult to dope cobalt ions into the crystal lattice of CsPbBr₃ nanocrystal through this method. Besides, the doping of Co has no influence on the crystalline structure of CsPbBr₃ (Figure S18) and the UV–vis spectra (Figure S19) of CsPbBr₃@GDY_{0.3}, and the HRTEM image of CsPbBr₃@GDY_{0.3}-Co (Figure 5a) shows no nanoparticles other than CsPbBr₃ nanocrystals. All these results suggest that the presence of Co in CsPbBr₃@GDY_{0.3}-Co is supported by GDY in a monodisperse form. As the reduction potential of Co²⁺/Co is around -0.76 V vs NHE,¹³ it is thermodynamically feasible to trap the photoexcited electrons in CsPbBr₃@GDY_{0.3}-Co and act as active sites to drive the conversion of CO₂-to-CO (-0.52 V vs NHE). Besides, the binding energy of Co 2p in CsPbBr₃@GDY_{0.3}-Co shows an apparent negative shift in comparison with CoBr₂; combined with the significant positive shifts of binding energies for Cs 3d, Pb 4f, Br 3d, and C 1s in CsPbBr₃@GDY_{0.3}-Co with respect to those in CsPbBr₃@GDY_{0.3} (Figures S20–23), the strong binding behavior of Co²⁺ on CsPbBr₃@GDY_{0.3} is evidenced, and the photoexcited electrons in CsPbBr₃@GDY_{0.3} are conducive to transfer to Co²⁺ sites. Furthermore, TEM (Figure 5a) and elemental mappings (Figure 5c–g) indicate that Co cations are

uniformly dispersed in CsPbBr₃@GDY_{0.3}-Co. Besides, CsPbBr₃@GDY_{0.3}-Co shows an adsorption capacity of 1.25 cm³ g⁻¹ toward CO₂ (Figure 4d), which is close to that of CsPbBr₃@GDY_{0.3}, suggesting that the coating of GDY was the main reason to enrich CO₂ in the photocatalytic system.

The photocatalytic activity of CsPbBr₃@GDY_{0.3}-Co was investigated in the same reaction conditions as CsPbBr₃ and CsPbBr₃@GDY_{0.3}. The results in Figure 5b show that, without adding extra sacrificial reductants, the cobalt-doped CsPbBr₃@GDY_{0.3} exhibits significant improvement in photocatalytic reduction of CO₂ to CO (27.7 $\mu\text{mol g}^{-1} \text{h}^{-1}$), 8 times higher than that of pristine CsPbBr₃ (3.3 $\mu\text{mol g}^{-1} \text{h}^{-1}$), further indicating the beneficial transfer of photogenerated electrons from CsPbBr₃@GDY_{0.3} to doping cobalt sites. In addition, the transient PL, photocurrent response, and EIS measurements (Figure 4a–c) concurrently indicated that the doping of cobalt sites can further improve the separation of photogenerated carriers and increase the number of active sites for CO₂ reduction. Furthermore, the decoration of Fe on CsPbBr₃@GDY_{0.3} (coded as CsPbBr₃@GDY_{0.3}-Fe) can also improve the activity of CsPbBr₃@GDY_{0.3} from 10.2 $\mu\text{mol g}^{-1} \text{h}^{-1}$ to 21.5 $\mu\text{mol g}^{-1} \text{h}^{-1}$ (Figure 5b), demonstrating that the coating of GDY can act as a compatible platform for decorating a variety of active sites for other catalytic reactions. To investigate the stability of CsPbBr₃@GDY_{0.3}-Co in the photoreaction system, the cycle experiments of CsPbBr₃@GDY_{0.3}-Co and CsPbBr₃ have been conducted. The results in Figure S24 show that there is 11% deactivation of CO production activity after four continuous cycles of the photoreactions for CsPbBr₃@GDY_{0.3}-Co. In contrast, the photocatalytic performance of CsPbBr₃ decreased by 58% after the same recycling experiments, demonstrating that CsPbBr₃@GDY_{0.3}-Co is more stable in the photoreaction system than CsPbBr₃. In addition, the varieties of structure, morphology, and electronic structure after cycles for CsPbBr₃@GDY_{0.3}-Co were also studied. As shown in Figures S25–S27, the TEM image, XRD, and Raman spectra of CsPbBr₃@GDY_{0.3}-Co demonstrate no significant morphologic and electronic structure changes after photocatalytic reactions. These results indicate that CsPbBr₃@GDY_{0.3}-Co was relatively stable compared to pure CsPbBr₃ during photocatalytic CO evolution, demonstrating that the coating of GDY

could improve the stability and the photocatalytic performance of CsPbBr₃-based catalysts. Furthermore, when the photocatalytic reactions were carried out on a large scale (with 21 mg catalyst), the conversion activities of CO₂-to-CO for CsPbBr₃@GDY_{0.3} (9.8 μmol g⁻¹ h⁻¹) and CsPbBr₃@GDY_{0.3}-Co (25.0 μmol g⁻¹ h⁻¹) are better than pure CsPbBr₃ (3.0 μmol g⁻¹ h⁻¹) (Figure S28). The highest CO evolution activity was also achieved by CsPbBr₃@GDY_{0.3}-Co, which is 8.3 times higher than that of pure CsPbBr₃, indicating that the as-prepared CsPbBr₃@GDY_{0.3}-Co photocatalyst possesses superior performance for photocatalytic reduction of CO₂ even at large scale. Finally, the proposed reaction pathways for the reduction of CO₂ to CO and the oxidation of H₂O were illustrated in Figure S29 referring to previous reports.^{48,49} For the reduction of CO₂, the Co²⁺ ions on CsPbBr₃@GDY_{0.3}-Co could act as the active site. With the supply of photoexcited electron (e⁻) on CsPbBr₃@GDY_{0.3}, the proton-coupled electron transfer (PCET) process and the cleavage of the C–OH bond within (CsPbBr₃@GDY_{0.3}-Co)···COOH can take place to generate CO. For the oxidation of H₂O, the (CsPbBr₃@GDY_{0.3}-Co)=O species can be formed after subtraction of the electrons, by photoexcited holes (h⁺), and protons. Then (CsPbBr₃@GDY_{0.3}-Co)=O could undergo the O–O coupling with water to generate (CsPbBr₃@GDY_{0.3}-Co)–O–OH, and produces oxygen through a reductive elimination step to regenerate the CsPbBr₃@GDY_{0.3}-Co.

3. CONCLUSIONS

In summary, we have successfully synthesized CsPbBr₃@GDY composite photocatalyst by in situ coating a 1.9 nm-thick GDY layer onto CsPbBr₃ nanocrystals. The thin GDY layer can not only enhance the stability of CsPbBr₃ in the water-containing system, but also acts as a hole-transfer layer to facilitate the transfer of photogenerated holes from CsPbBr₃ to GDY, thus suppressing the electron–hole recombination. Furthermore, benefiting from the high loading capacity of metal ions, the GDY layer can serve as a compatible platform for decorating Co²⁺ to catalyze the reduction of CO₂. After doping of Co ions, the photocatalytic activity of CsPbBr₃@GDY_{0.3}-Co for the CO₂-to-CO conversion reaches up to 27.7 μmol g⁻¹ h⁻¹, which is almost 8 times higher than that of pure CsPbBr₃. The GDY-protected CsPbBr₃, suppressed electron–hole recombination, doped Co active sites, and improved chemisorption of CO₂ on GDY are responsible for the enhancement of photocatalytic CO₂-to-CO reduction. This work provides a fascinating strategy for improving the stability and activity of metal-halide perovskite nanocrystals and broadens their applications in the field of energy conversion.

4. EXPERIMENTAL SECTION

4.1. Synthesis of CsPbBr₃@GDY. First, hexakis[(trimethylsilyl)ethynyl]benzene (HEB-TMS) (1 mg, 3 mg, or 5 mg) was dissolved in tetrahydrofuran (5 mL) in a three-necked flask under Ar atmosphere at 273 K in the dark, followed by the quick addition of tetra-*n*-butylammonium fluoride (TBAF) (20 μL, 60 or 100 μL) into the above solution, after stirring for 10 min, hexakis(ethynyl)benzene (HEB) (0.33 mg, 0.99 mg, or 1.65 mg) was obtained by extraction operation. Second, the obtained HEB was dissolved in a mixed solvent of toluene and *n*-hexane (1:1, *v:v*) for use. Then, the as-prepared CsPbBr₃ (10 mg, see details in the SI) was added to the above solutions, and the reaction system was transferred to a microwave synthesizer and subjected to microwave irradiation for 10 min. Finally, the synthesized samples were washed, dried, and then stored in a vacuum desiccator for use. CsPbBr₃@GDY with different compound

ratios were labeled as CsPbBr₃@GDY_{0.1}, CsPbBr₃@GDY_{0.3}, and CsPbBr₃@GDY_{0.5}, respectively.

4.2. Synthesis of CsPbBr₃@GDY_{0.3}-Co. The synthesis process of CsPbBr₃@GDY_{0.3}-Co is similar to that of CsPbBr₃@GDY_{0.3}, except that CoBr₂ (1 mg) and CsPbBr₃ (10 mg) were added to HEB mixture solution at the same time.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsami.0c14826>.

Preparation process of CsPbBr₃; the characterization of CsPbBr₃, CsPbBr₃@GDY, and CsPbBr₃@GDY-Co; and additional results of TEM, Raman spectra, XRD spectra, GC–MS analysis, UV–vis spectrum, Mott–Schottky plots, and XPS measurements (PDF)

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Notes

The authors declare no competing financial interest.

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